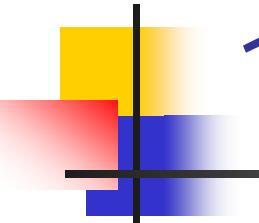


Core loss spectra (EELS, XAS)

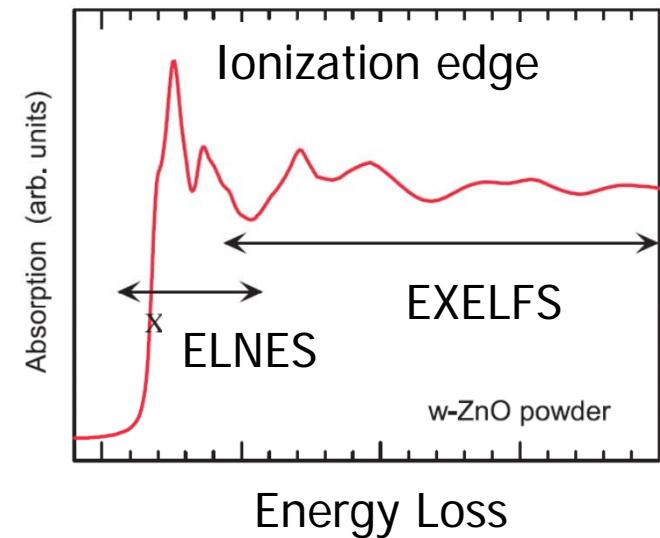
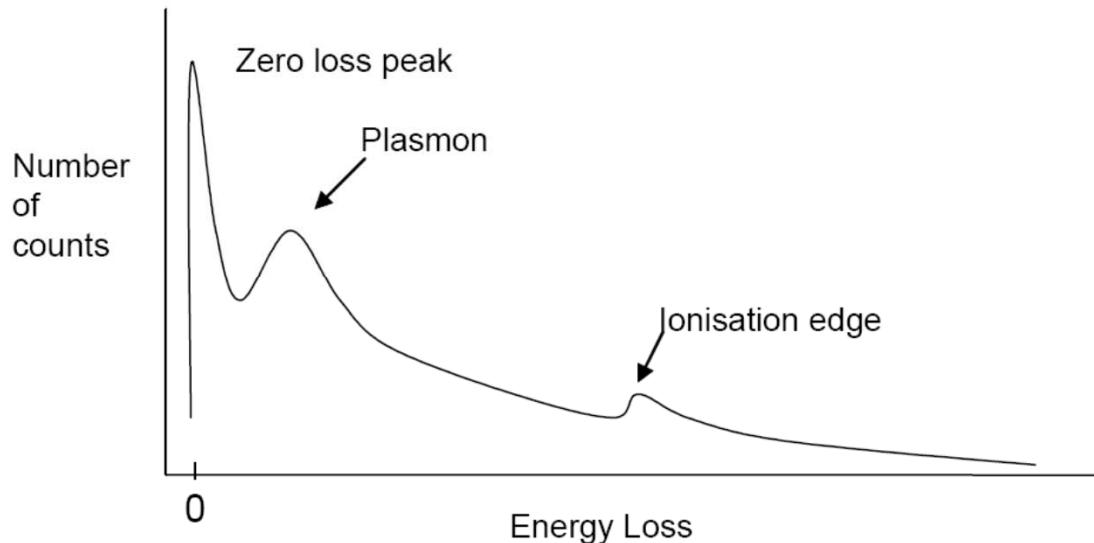
Kevin Jorissen
University of Washington (USA)
WIEN2k 2013 Penn State



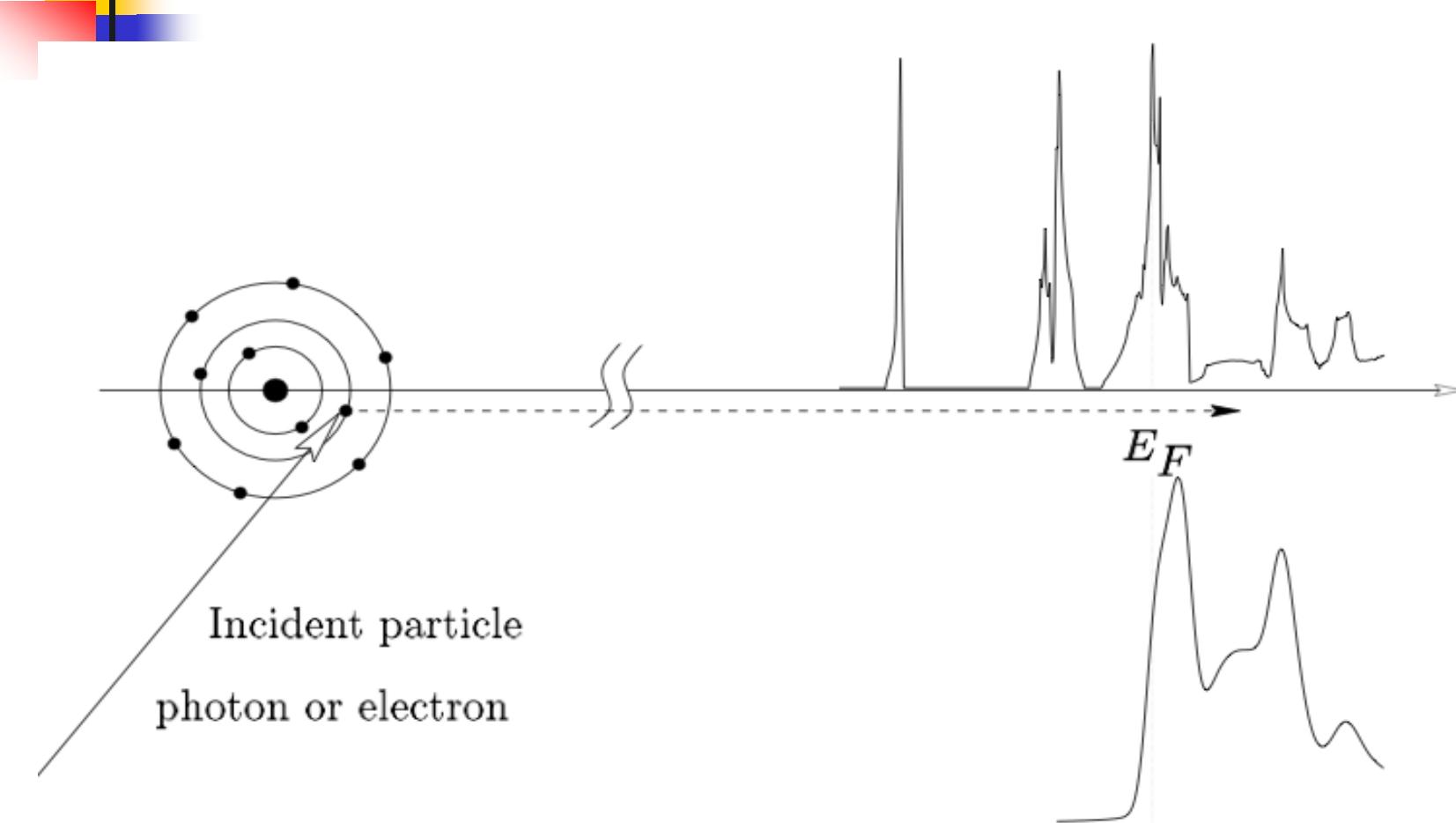
1. Concepts

WIEN2k calculates ELNES / XANES

- EELS : Electron Energy Loss Spectroscopy
- XAS: X-ray Absorption Spectroscopy

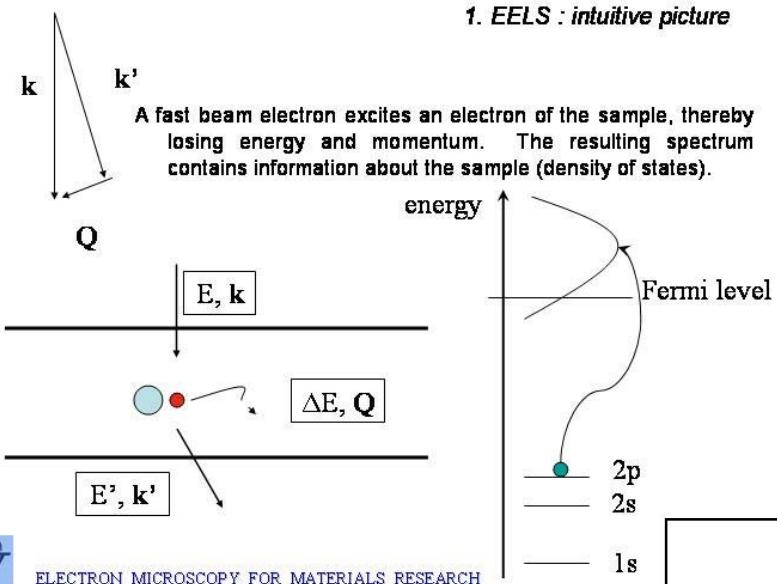
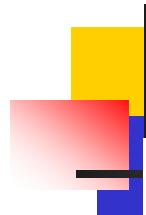


the excitation process



INTRODUCING EELS

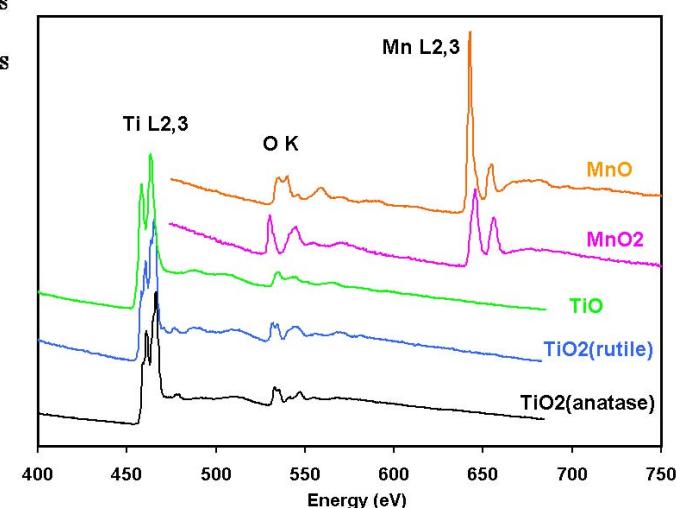
Electron Energy Loss Spectroscopy is performed in a Transmission Electron Microscope, using a beam of high-energy electrons as a probe. The energy distribution of the beam gives a loss spectrum similar to XAS. Focussed probes give excellent spatial resolution ($\sim 0.5 \text{ \AA}$).
Energy resolution is improving ($\sim 25 \text{ meV}$).



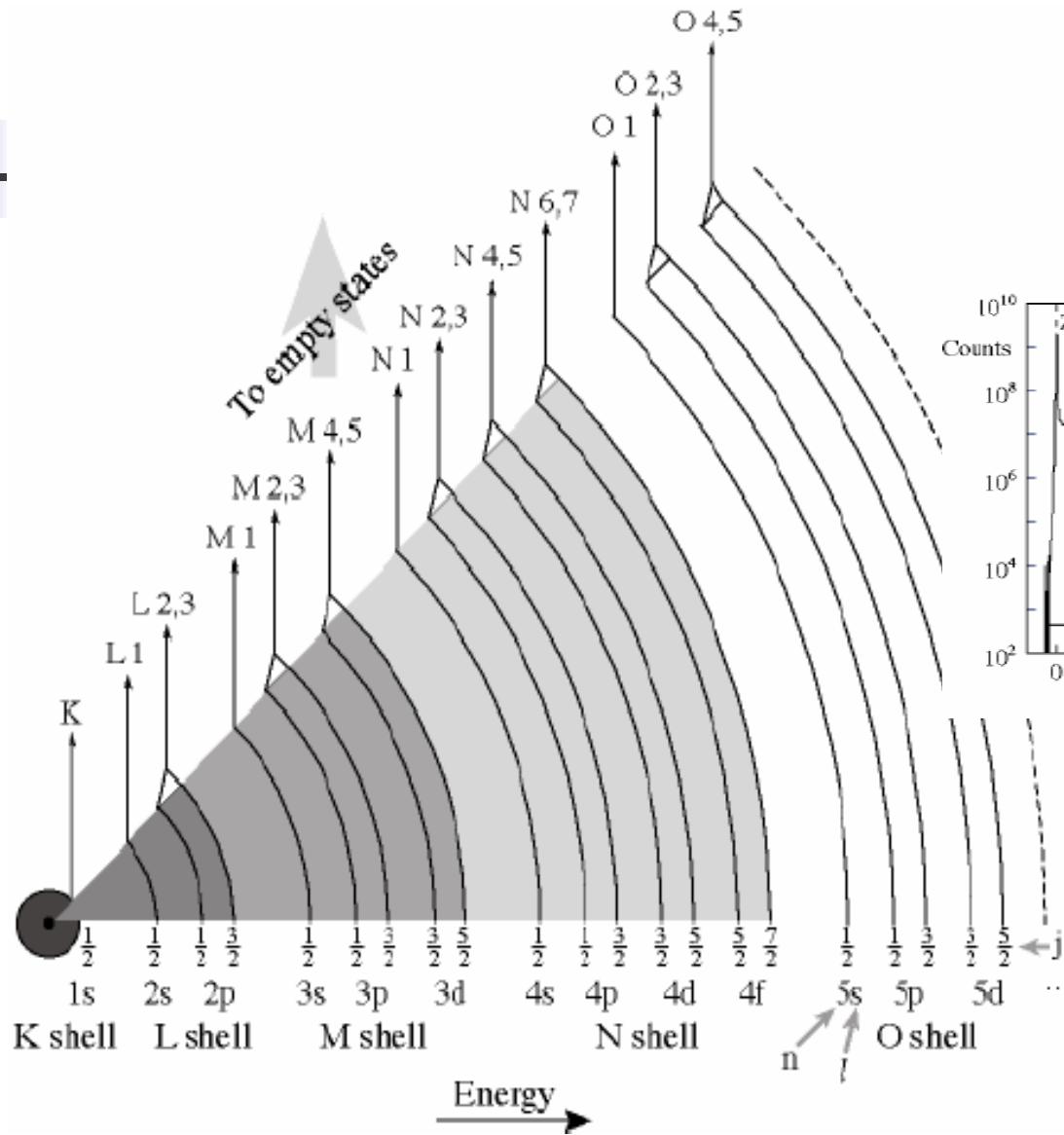
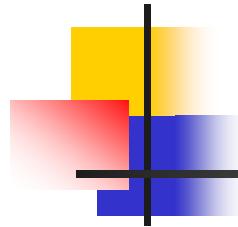
Intuitive picture of EELS

EELS spectra of TM oxides
Probes local electronic structure

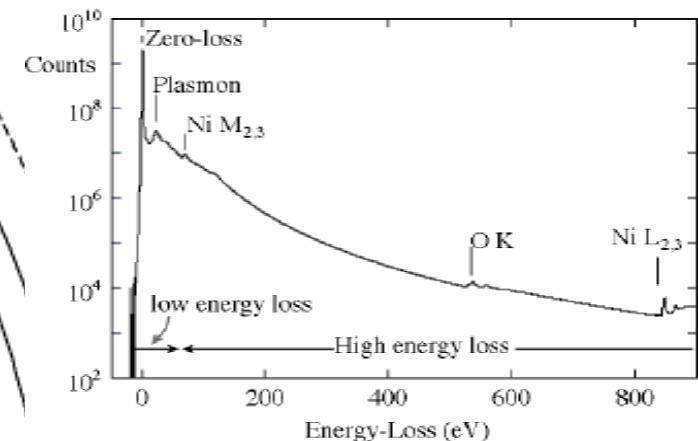
Electron microscope equipped with EELS-detector



Terminology for ionization edges



Inner shell ionization.



instrumentation

XAS: synchrotron



EELS: microscope



THEORY OF EELS : A double differential scattering cross-section is calculated by summing over all possible transitions between initial and final states.

The transition probabilities are described by Fermi's golden rule.

$$\frac{\partial^2 \sigma}{\partial E \partial \Omega}(E, Q) = \zeta \sum_{I,F} \frac{k_F}{k_I} \left| \langle I k_I | V | k_F F \rangle \right|^2 \delta(E_I - E_F)$$

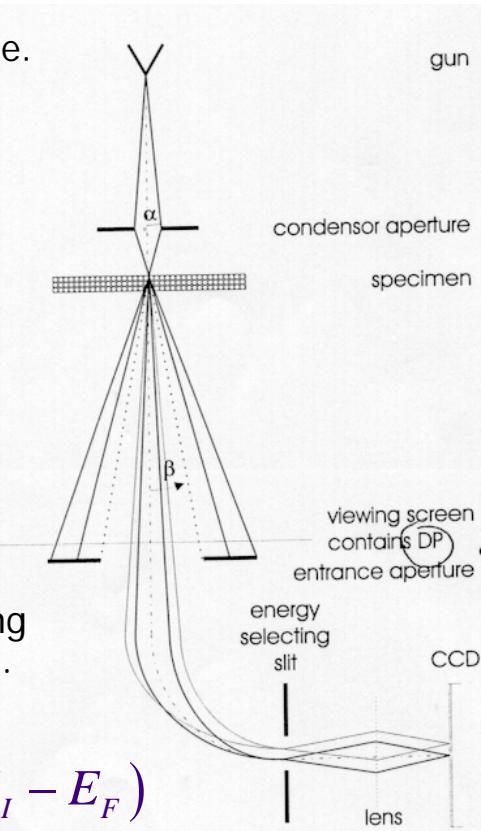
V is the interaction potential between the fast beam electron and an electron in the sample.

F, I the sample states, can be taken from electronic structure calculations.

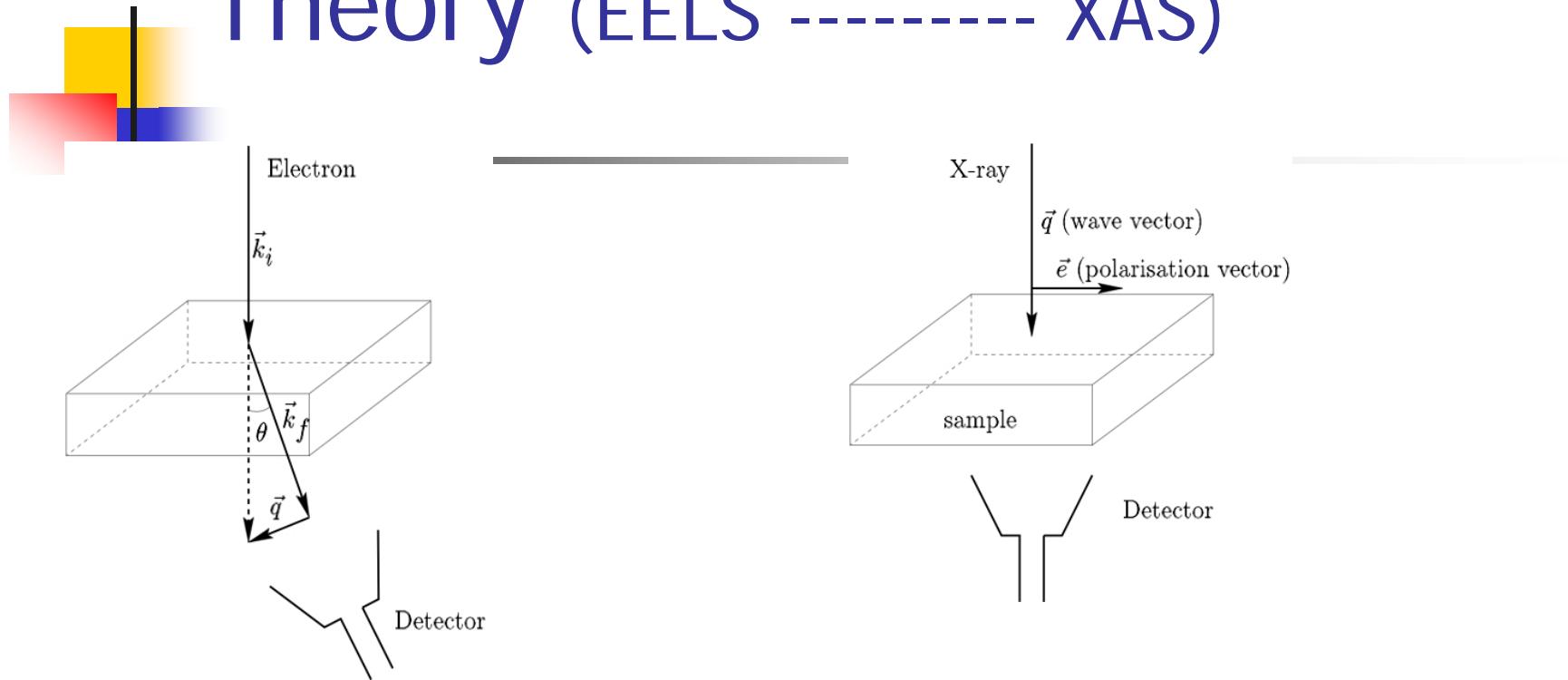
k_F and k_I the probe states, are typically described as plane waves when Bragg scattering effects are neglected.

In experiment, one usually integrates over a range of scattering angles, due to the beam width and spectrometer aperture.
 → differential cross section :

$$\frac{\partial \sigma}{\partial E}(E; \alpha, \beta) = \int_{\alpha, \beta} d\Omega \zeta \sum_{I,F} \frac{k_F}{k_I} \left| \langle I k_I | V | k_F F \rangle \right|^2 \delta(E_I - E_F)$$



Theory (EELS ----- XAS)



$$\frac{\partial^2 \sigma}{\partial E \partial \Omega} \propto \sum_{I,F} \left| \langle I | e^{i\vec{q} \cdot \vec{R}} | F \rangle \right|^2$$

$$\frac{\partial \sigma}{\partial E} \propto \sum_{I,F} \left| \langle I | e^{i\vec{q} \cdot \vec{R}} \vec{e} \cdot \vec{R} | F \rangle \right|^2$$

dipole approximation

$$\vec{q} \cdot \vec{R} \ll 1 \rightarrow e^{i\vec{q}\vec{R}} = 1 + i\vec{q} \cdot \vec{R} + \frac{(\vec{q} \cdot \vec{R})^2}{2!} + \square$$

EELS

$$\frac{\partial^2 \sigma}{\partial E \partial \Omega} \propto \sum_{I,F} \left| \langle I | \vec{q} \cdot \vec{R} | F \rangle \right|^2$$

XAS

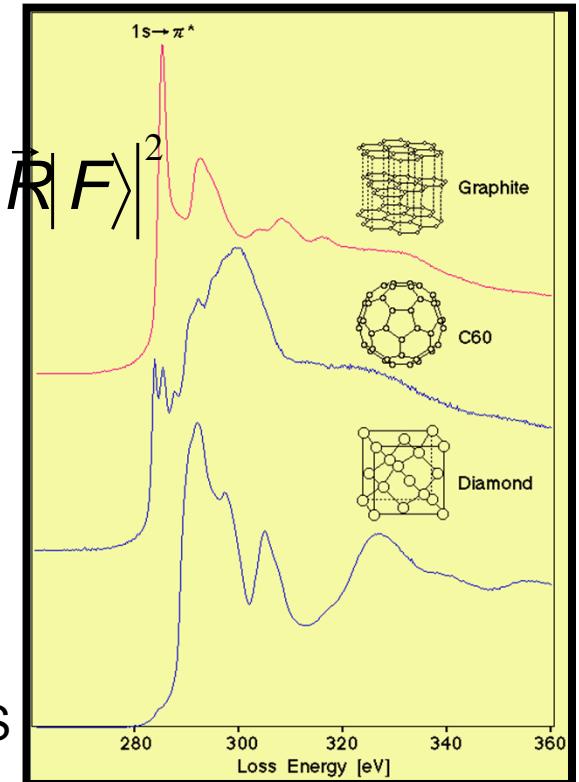
$$\frac{\partial^2 \sigma}{\partial E \partial \Omega} \propto \sum_{I,F} \left| \langle I | \vec{e} \cdot \vec{R} | F \rangle \right|^2$$

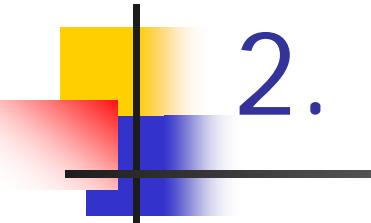
The polarization vector e in XAS plays the same role as momentum transfer q in ELNES within the dipole approximation.

This is why people say “XAS = EELS”.

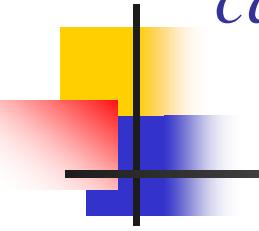
(Beware - there are quite a few differences, too.)

→ Probes local, symmetry-selected (I_c+1) unoccupied DOS





2. WIEN2k Calculations.



calculation of spectra using WIEN2k

Set up structure and initialize

SCF calculation

x qtl -telnes

Prepare case.innes

Prepare case.inxs

or

x telnes3

x xspec

↑
EELS

↑
XAS

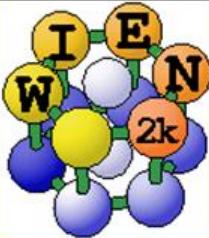
x broadening

ELNES workflow

Cr3C2@raphael.phys.washington.edu - Windows Internet Explorer
http://raphael.phys.washington.edu:7890/index.pl?SID=954086

Favorites Gmail - Inbox - kevinjoriss... DS De Standaard Online Cr3C2@raphael.phys.w... Page Safety Tools >>

Session: **Cr3C2**
[/phys/users/jorissen/Cr3C2](#) 13:35:44 idle
[refresh](#) | [no refresh](#)



TELNES3

[edit Cr3C2.innes](#) Edit input-file for ELNES (InnesGen™)

Only if you want to include states with higher energy

[edit Cr3C2.in1](#) Edit in1

[x lapw1](#) Calculate eigenvalues interactively

[x qtl-telnes](#) Calculate partial charges interactively

[x telnes3](#) Calculate ELNES spectra interactively

[view Cr3C2.outputelnes](#) display Cr3C2.outputelnes (optional)

[edit Cr3C2.inb](#) Edit input-file for BROADENING

[x broadening](#) Broaden the spectrum interactively

[plot](#) Plot ELNES

[save_eels](#) Save an elnes calculation into a directory

Execution >>

- [StructGen™](#)
- [initialize calc.](#)
- [run SCF](#)
- [single prog.](#)
- [optimize\(V,c/a\)](#)
- [mini_positions](#)

Utils. >>

- [**<< Tasks**](#)
- [El_Dens.](#)
- [DOS](#)
- [XSPEC](#)
- [TELNES3](#)
- [OPTIC](#)
- [Bandstructure](#)

Files >>

- [struct file\(s\)](#)
- [input files](#)
- [output files](#)
- [SCF files](#)

Session Mgmt. >>

- [change session](#)
- [change dir](#)
- [change info](#)

Configuration

Usersguide
Implementation and realization by
Joris Jorissen 2001-2006

ELNES input w2web

Cr3C2@raphael.phys.washington.edu - Windows Internet Explorer
http://raphael.phys.washington.edu:7890/index.pl?SID=954086

Favorites Gmail - Inbox - kevinjoriss... DS De Standaard Online Cr3C2@raphael.phys.w... Page Safety Tools >> 13:31:13 idle refresh | no refresh

Session: Cr3C2 /phys/users/jorissen/Cr3C2



Title: Cr L1 edge of first atom

Atom: 1: Cr0+ **Edge:** L1 (n=1 l=0)

Edge onset: 696 eV **Beam energy:** 200 keV

Energy grid: 0.0000 eV to 15.0000 eV in steps of 0.0500 eV

Collection s.a.: 5.00 mrad **Convergence s.a.:** 1.87 mrad

Spectrometer broadening: 0.50 eV **Q-mesh:** NR=5 NT=2

Advanced settings:

Branching ratio: [input field] (statistical if empty)

Spinorbit splitting of core state (eV): [input field] (calculated if empty)

Orientation sensitive: $\alpha = 0^\circ$, $\beta = 90^\circ$, $\gamma = 0^\circ$

Integrate over equivalent atoms: [input field] to [input field] (all eq. atoms if empty)

Detector position: θ_x 0.000 mrad, θ_y 0.000 mrad

Modus: energy

Initialization: Calculate DOS write DOS
 Calculate rotation matrices write rotation matrices

Verbosity: basic **File headers:** Write headers (default)

Interaction potential: relativistic (recommended)

Q-grid: U uniform $\theta_0 =$ [input field] (not used for uniform grid)

Interaction order: all & λ (default) **Final state selection rule:** $L=I \pm 1$ (default)

Extend potential beyond Rmt: $r_{max} =$ [input field] bohr

Set Fermi energy manually: $E_F =$ [input field] Ry

Read core state wavefunction: filename= case.cwf

Read final state wavefunctions: filename= case.finalwf

Calculate DOS only

Execution >>
StructGen™
initialize calc.
run SCF
single prog.
optimize(V.c/a)
mini_positions

Utils. >>
<< Tasks
El_Dens.
DOS
XSPEC
TELNES3
OPTIC
Bandstructure

Files >>
struct file(s)
input files
output files
SCF files

Session Mgmt. >>
change session
change dir
change info

Configuration

Usersguide
Version 1.0
Last updated by
Joris Jorissen 2001-2006

ELNES input file (case.innes)

Cr3C2@raphael.phys.washington.edu - Windows Internet Explorer
http://raphael.phys.washington.edu:7890/index.pl?SID=954086

Favorites Gmail - Inbox - kevinjoriss... DS De Standaard Online Cr3C2@raphael.phys.w... Page Safety Tools >>

Session: Cr3C2 /phys/users/jorissem/Cr3C2 13:40:56 idle refresh | no refresh



Execution >>

- StructGen™
- initialize calc.
- run SCF
- single prog.
- optimize(V.c/a)
- mini_positions

Utils. >>

<< Tasks

- EI_Dens.
- DOS
- XSPEC
- TELNES3
- OPTIC
- Bandstructure

Files >>

- struct file(s)
- input files
- output files
- SCF files

Session Mgmt. >>

- change session
- change dir
- change info

Configuration

Usersguide
Last update and realization by luitz at 07/01/2006

File: /phys/users/jorissem/Cr3C2/Cr3C2.innes

Save Download this file

```
Graphite C K edge of first atom.  
1  
1 0  
285.00  
300  
0.0000 15.0000 0.0500  
5.00 1.87  
5 2  
0.50  
DETECTOR POSITION  
0.000 0.000  
MODUS  
energy  
SELECTION RULE  
n  
LSELECTION RULE  
d  
INITIALIZATION  
Y Y  
Y Y
```

delete this file

w2web © luitz.at

2web



XSPEC-task



Session: [magnetite]
/area51/plaha/lapw/correlated/magnetite

16:42:50 1de
[refresh] | [no refresh]



[Execution >>]
[StructGen™]
[view structure]
[initialize calc.]
[run SCF]
[single prog.]
[optimize(V,c/a)]
[mini. positions]

[Utils. >>]

[<< Tasks]
[El. Dens.]
[DOS]
[XSPEC]
[TELNES.2]
[OPTIC]
[Bandstructure]

[Files >>]
[struct file(s)]
[input files]
[output files]
[SCF files]

[Session Mgmt. >>]
[change session]
[change dir]
[change info]

[Configuration]

Usersguide
[html-Version]
[pdf-Version]

XSPEC

[Spin UP] [Spin DOWN]

Spin UP selected.

If you want to include states with higher energy

edit magnetite.in1 Edit in1

x lapw1 -up Calculate eigenvalues interactively

x lapw1 -dn Calculate eigenvalues interactively

x lapw2 -qtl -up Calculate partial charges interactively

edit magnetite.inxs Edit input-file for XSPEC

x xspec -up Calculate X-ray spectra interactively

plot Plot XSPEC

Title: Atom 1 L3 absorption spectrum
1 (atom)
2 (n core)
1 (l core)
0,0.5,0.5 (split, Int1, Int2)
-2,0.02,15 (EMIN,DE,EMAX)
ABS (type of spectrum)
1.00 (S)
2.0 (gamma0)
1.50 (W only for EMIS)
AUTO (AUTO or MANually select Energy)
-6.93
-10.16
-13.9



Practical considerations

- Spectra usually converge easily with respect to RKMAX, k-mesh, SCF criteria
- But you should check anyway (see Cu L3)
- Optimizing positions may be necessary
- You may need to sum over all "C" atoms in the unit cell. (Especially for orientation-resolved calculations.)
- You probably need to use a "core hole". This can be a lot of work.
- Your results may be wrong even if you do everything right. (But often they are reasonably good.)
- To compare to experiment, you'll probably fiddle with the broadening, the onset energy, and the branching ratio (L3/L2)

Convergence of Cu L₃ edge with # k-points

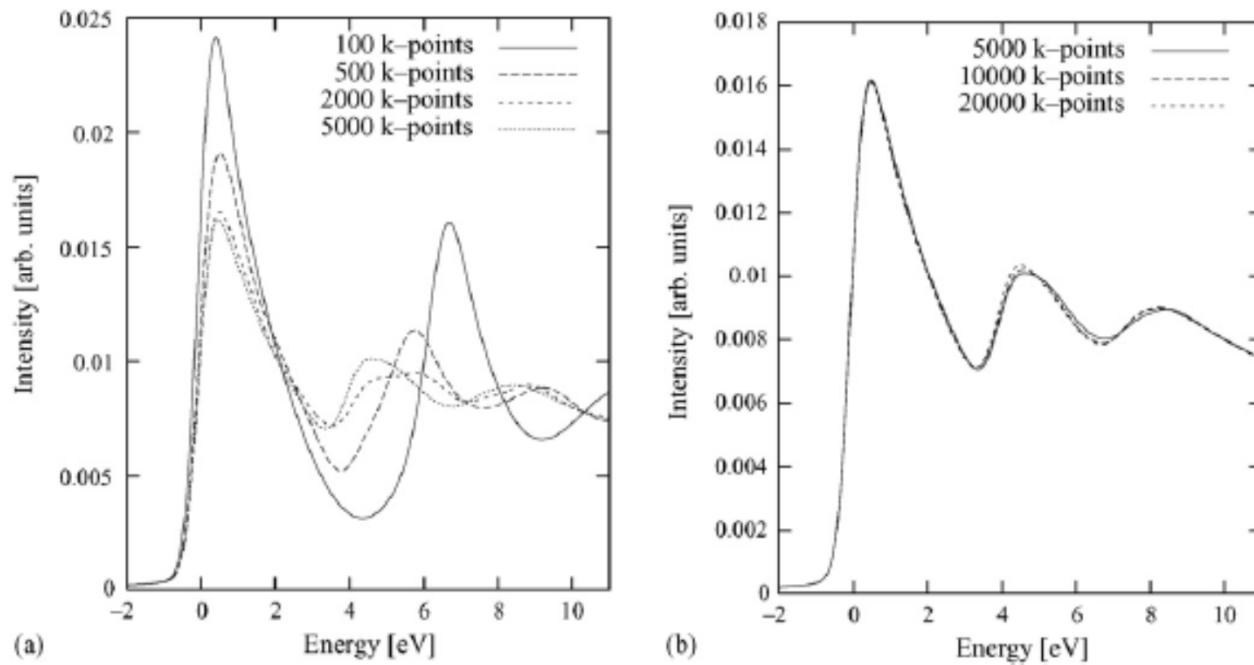
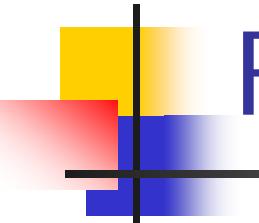


Fig. 8. Cu L₃ edge in fcc-Cu calculated for different number of k -points in the whole BZ. The calculation was performed with RKMAX = 8. The structures were broadened with a Gaussian of 0.7 eV to account for experimental broadening. Lifetime broadening was modeled with a linear approximation.



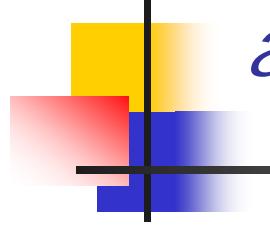
Features of WIEN2k

- Orientation dependence
- Beyond dipole selection rule
- Several broadening schemes
- All-electron

For EELS:

- Account for collection/convergence angle
- Output $\sigma(E)$ or $\sigma(\theta)$
- Relativistic ELNES (\rightarrow anisotropic materials)

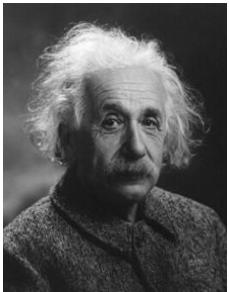
EELS – Relativistic theory needed for anisotropic materials



Semi-relativistic theory :

$$\frac{\partial^2 \sigma}{\partial E \partial \Omega}(E, \mathbf{Q}) = \frac{4\gamma^2}{a_0^2} \frac{k'}{k} \frac{1}{Q^4} \sum_{I,F} \left| \langle I | \mathbf{Q} \cdot \mathbf{r} | F \rangle \right|^2 \delta(E_I - E_F - E)$$

$$\begin{aligned} V &= |\mathbf{r} - \mathbf{r}'|^{-1} \\ m &\rightarrow \gamma m \\ \theta_E &\rightarrow \theta_{E,\text{rel}} \end{aligned}$$



Fully relativistic theory (P. Schattschneider et al., Phys. Rev. B 2005) :

Up to leading order in c^{-2} and using the Lorentz gauge : $V = e\Phi \left(1 - \frac{\mathbf{p} \cdot \mathbf{v}_0}{mc^2} \right)$ $\Phi = \frac{-4\pi e \delta(\omega - \mathbf{q} \cdot \mathbf{v}_0)}{q^2 - \omega^2/c^2}$

$$\frac{\partial^2 \sigma}{\partial E \partial \Omega}(E, \mathbf{Q}) = \zeta \frac{k'}{k} \frac{1}{(Q^2 - E/\hbar c)^2} \sum_{I,F} \left| \langle I | \mathbf{r} \cdot (\mathbf{Q} - Q_z \beta^2 \mathbf{e}_z) | F \rangle \right|^2 \delta(E_I - E_F - E)$$

Geometrical interpretation : in the dipole limit, a relativistic Hamiltonian shrinks the impuls transfer in the direction of propagation. (The general case is more complex.)

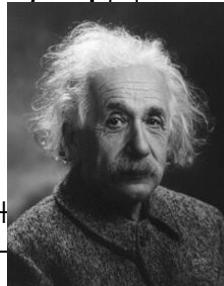
WIEN2k can also calculate non-dipole relativistic transitions. The equations are so long they make PowerPoint cry.

Beyond the small q approximation

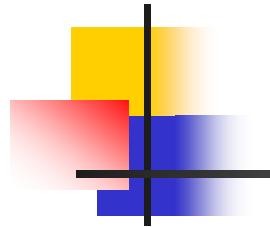
The relativistic DDSCS :

$$\frac{\partial^2 \sigma}{\partial E \partial \Omega} = \left[\frac{4\gamma^2 a_0^{-2}}{q^2 - (E/\hbar c)^2} \right]^2 \frac{k_f}{k_i} \sum_{i,f} \left| \left\langle f \left| e^{i\mathbf{q} \cdot \mathbf{r}} \left(1 - \frac{\mathbf{v}_0 \cdot \mathbf{p}}{m_e c^2} \right) \right| i \right\rangle \right|^2 \delta(E_f - E_i - E)$$

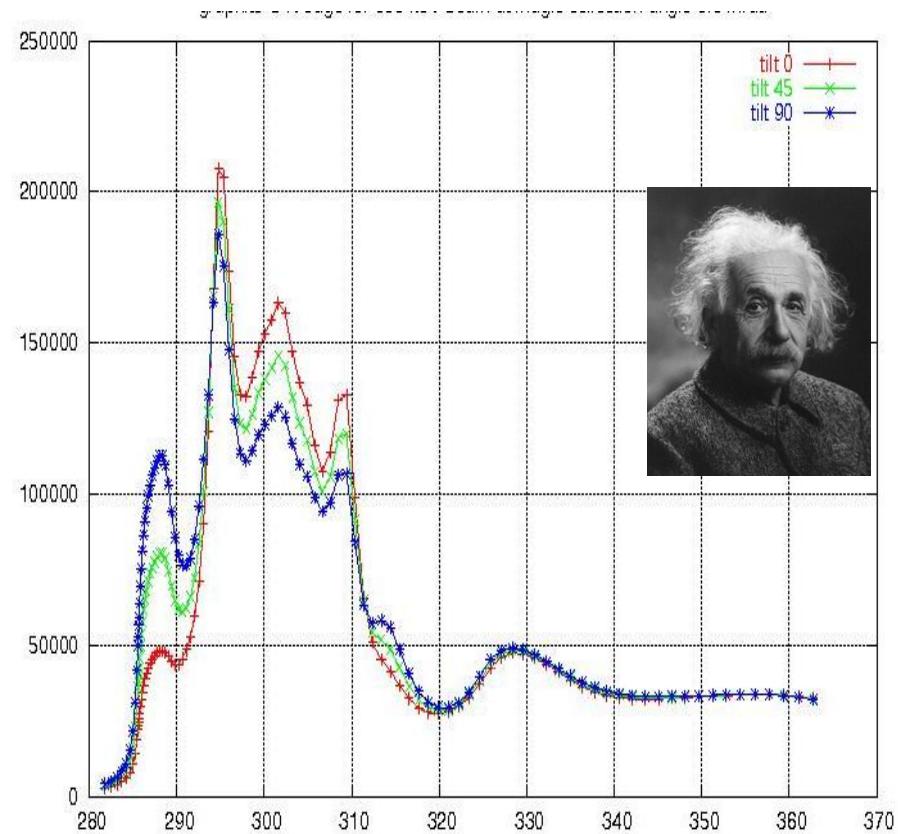
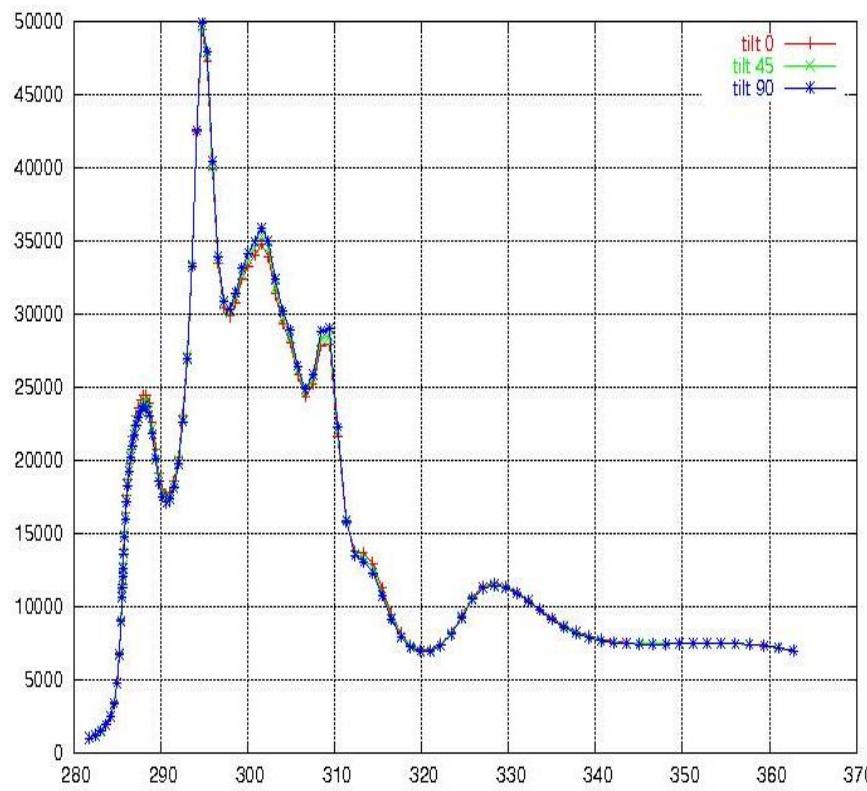
$$\begin{aligned}
 \langle i | V | f \rangle &= 4\pi \sum_{\lambda\mu} \sum_{lm} i^\lambda d_{lm}^{f*} Y_{\lambda\mu}^*(\Omega_q) \left[t_1 + i \sum_{a=2}^5 t_a \right] \\
 &= 4\pi \sum_{\lambda\mu} \sum_{lm} i^\lambda d_{lm}^{f*} Y_{\lambda\mu}^*(\Omega_q) \left\{ \begin{Bmatrix} l & \lambda & l_i \\ -m & \mu & m_i \end{Bmatrix} \int j_\lambda u_l u_i + \frac{i\hbar v_0}{m_e c^2} \sqrt{\frac{4\pi}{3}} \int j_\lambda u_l \left(\frac{\partial u_i}{\partial r} - \frac{m_i}{r} u_i \right) \right. \\
 &\quad \left[\begin{Bmatrix} l_i+1 & l_i & 1 \\ -m_i & m_i & 0 \end{Bmatrix} \begin{Bmatrix} l & \lambda & l_i+1 \\ -m & \mu & m_i \end{Bmatrix} + \begin{Bmatrix} l_i-1 & l_i & 1 \\ -m_i & m_i & 0 \end{Bmatrix} \begin{Bmatrix} l & \lambda & l_i-1 \\ -m & \mu & m_i \end{Bmatrix} \right. \\
 &\quad \left. \int j_\lambda u_l \frac{u_i}{r} \left[\begin{Bmatrix} l_i+1 & l_i & 1 \\ -m_i & m_i+1 & -1 \end{Bmatrix} \begin{Bmatrix} l & \lambda & l_i+1 \\ -m & \mu & m_i \end{Bmatrix} + \begin{Bmatrix} l_i-1 & l_i & 1 \\ -m_i & m_i+1 & -1 \end{Bmatrix} \begin{Bmatrix} l & \lambda & l_i-1 \\ -m & \mu & m_i \end{Bmatrix} \right] \right] \right\} \\
 &\quad \left. \int j_\lambda u_l \frac{u_i}{r} \left[\begin{Bmatrix} l_i+1 & l_i & 1 \\ -m_i & m_i+1 & -1 \end{Bmatrix} \begin{Bmatrix} l & \lambda & l_i+1 \\ -m & \mu & m_i \end{Bmatrix} + \begin{Bmatrix} l_i-1 & l_i & 1 \\ -m_i & m_i+1 & -1 \end{Bmatrix} \begin{Bmatrix} l & \lambda & l_i-1 \\ -m & \mu & m_i \end{Bmatrix} \right] \right]
 \end{aligned}$$



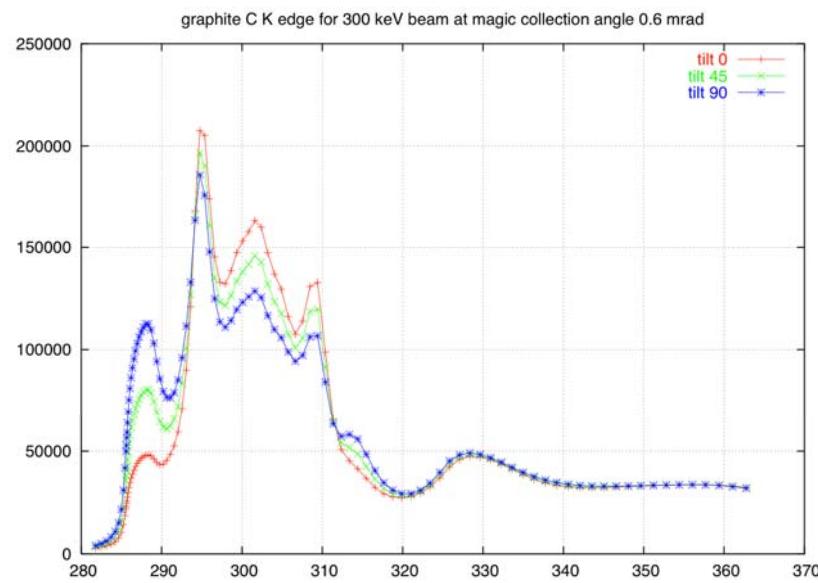
Relativistic spectra



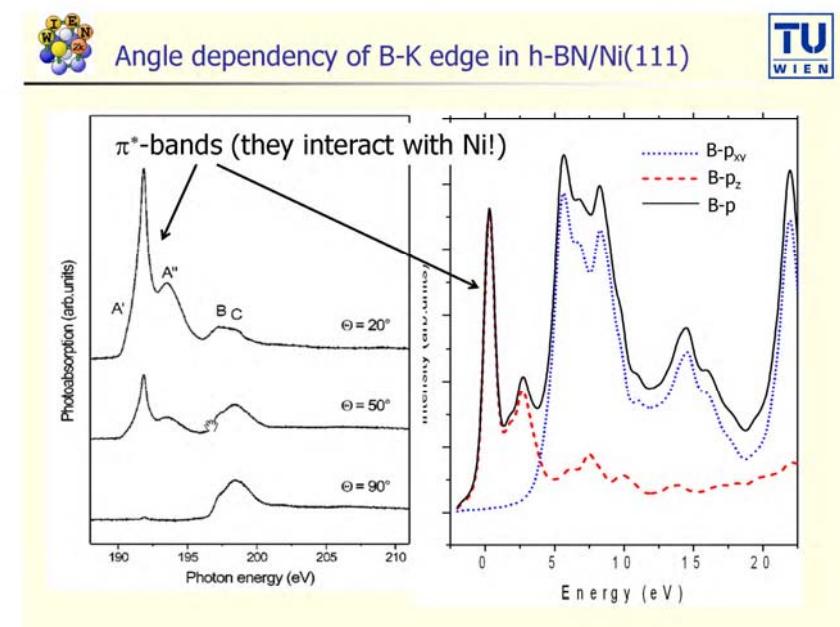
Graphite C K for 3 tilt angles. Beam energy 300 keV, collection angle = 2.4mrad.
Left: nonrelativistic calculation. Right: relativistic calculation.



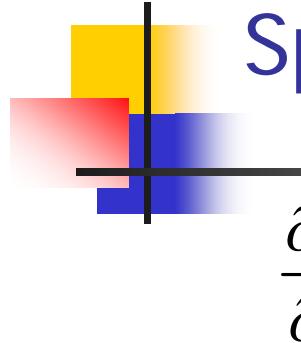
Orientation dependence



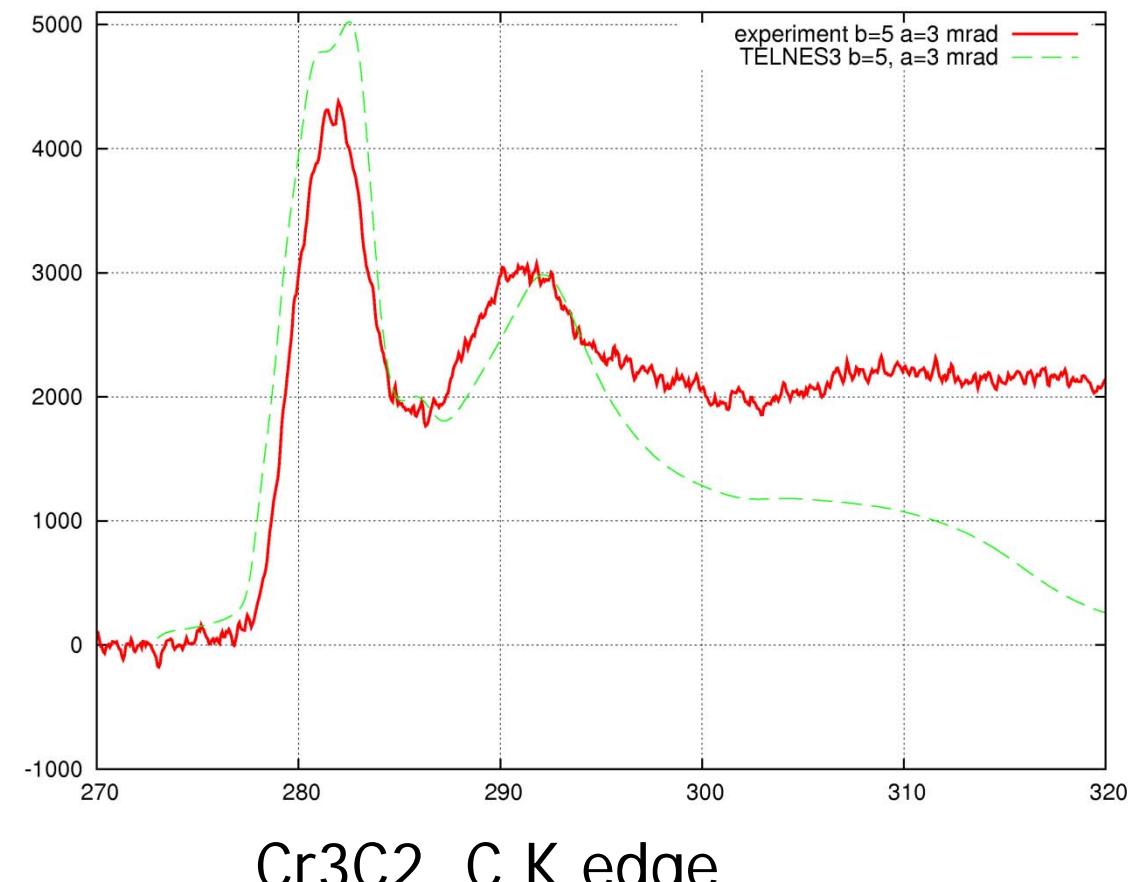
graphite C K EELS



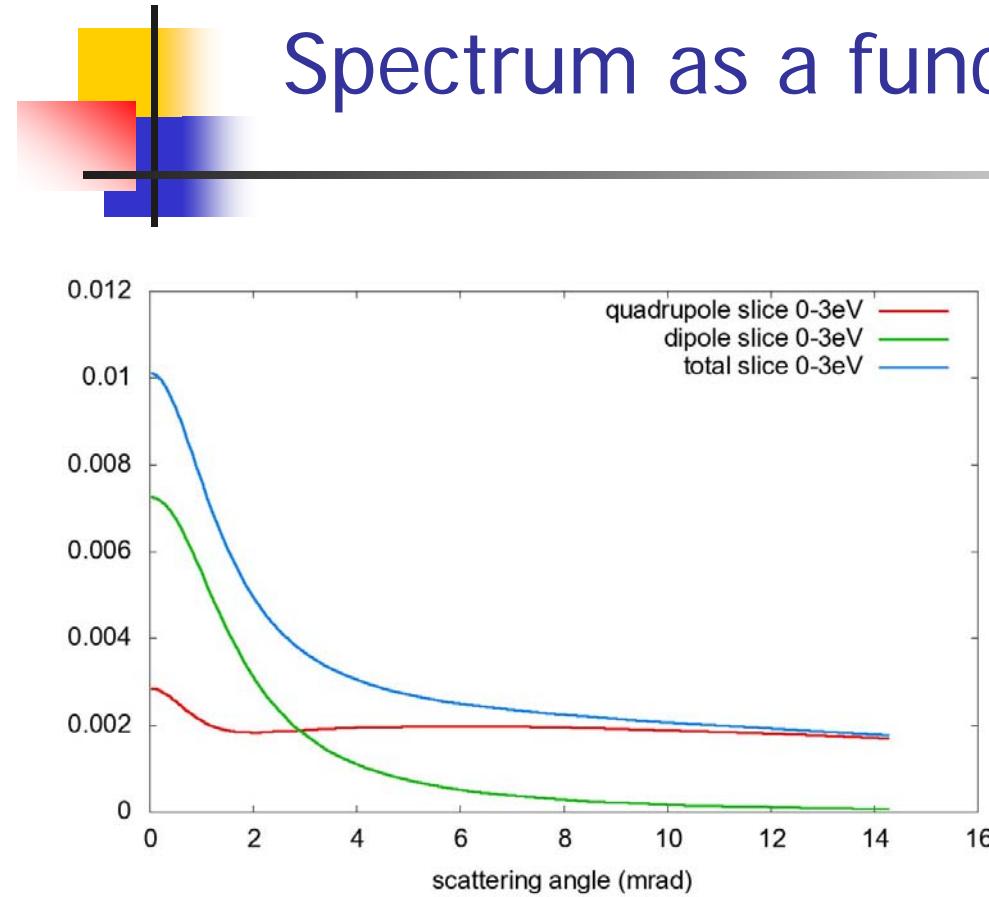
BN B K XAS



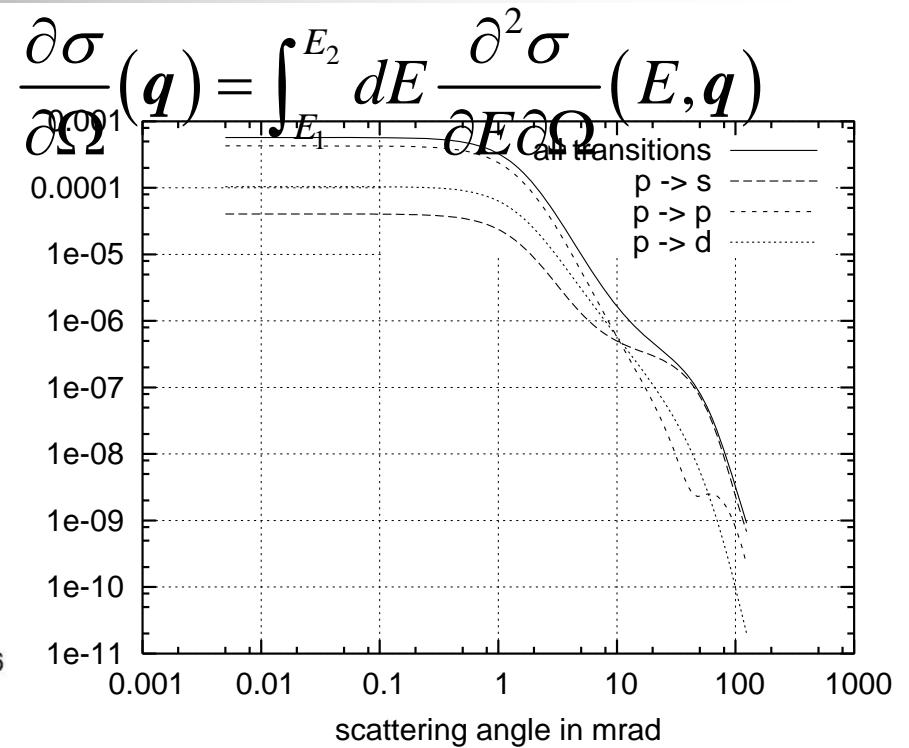
Spectrum as a function of energy loss



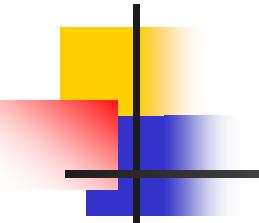
Spectrum as a function of scattering angle



Left : L3 edge of Cr₃C₂

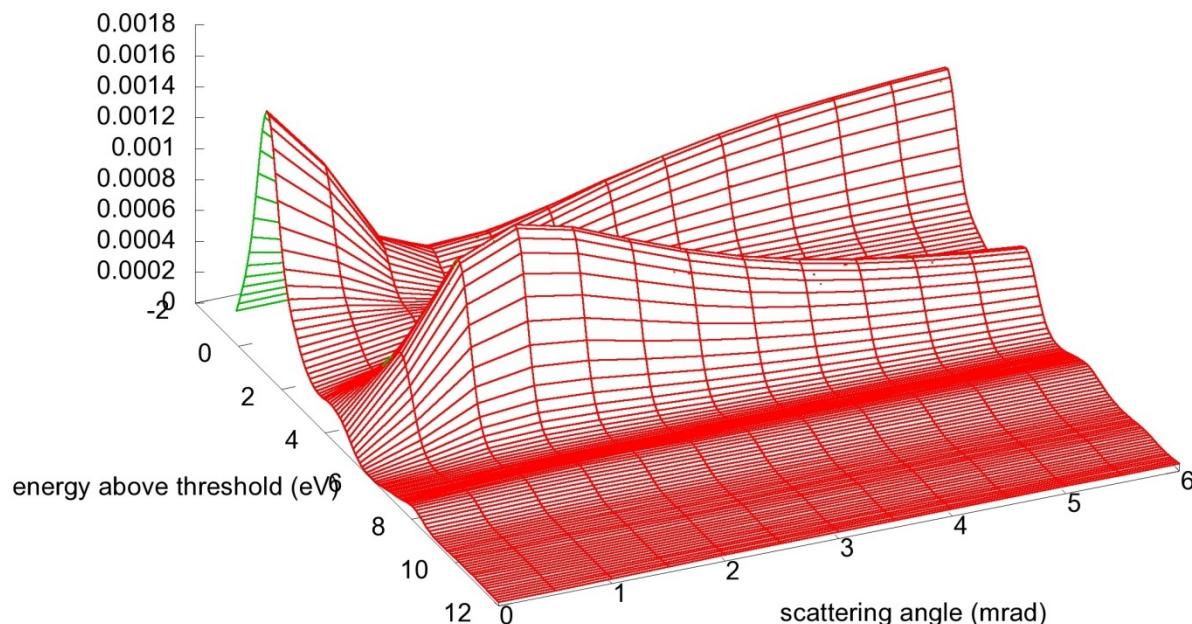


Right : the As L3 edge of NiAs (1324 eV)
Calculated using WIEN2k+TELNES2



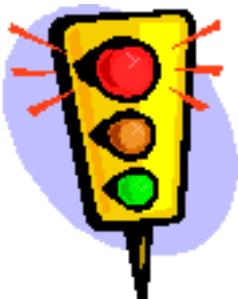
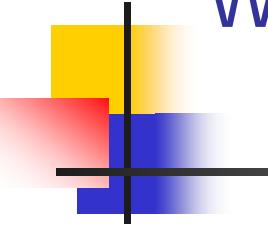
Just the double-differential CS

Double differential scattering cross-section (DDSCS)



$$\frac{\partial^2 \sigma}{\partial E \partial \Omega}(E, q)$$

Warning!

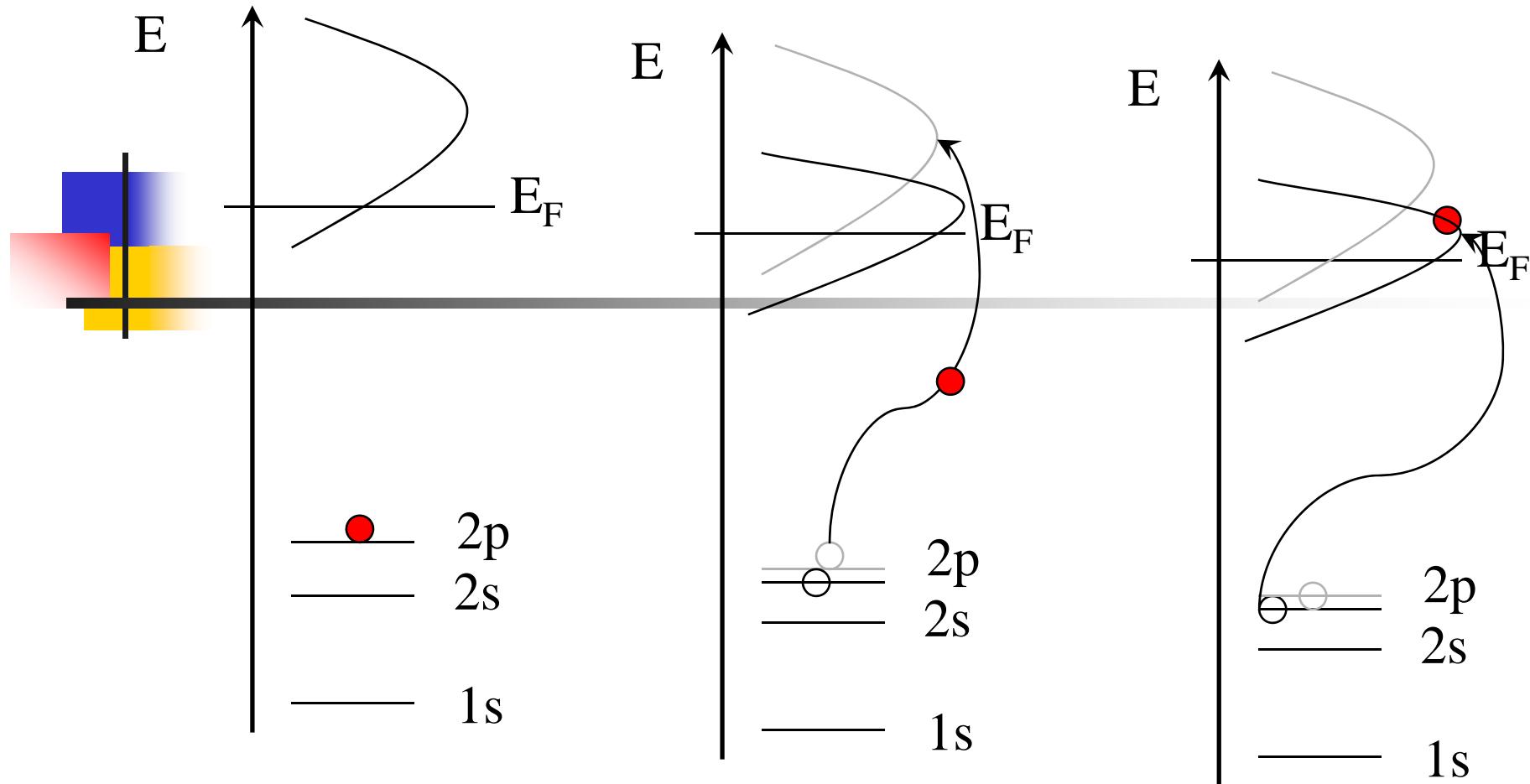


✗ DFT is a ground state theory !

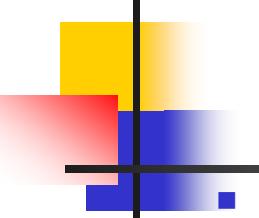
→ it should fail for the prediction of excited state properties

✓ *however: for many systems it works pretty well*

The core hole



ELECTRON MICROSCOPY FOR MATERIALS RESEARCH

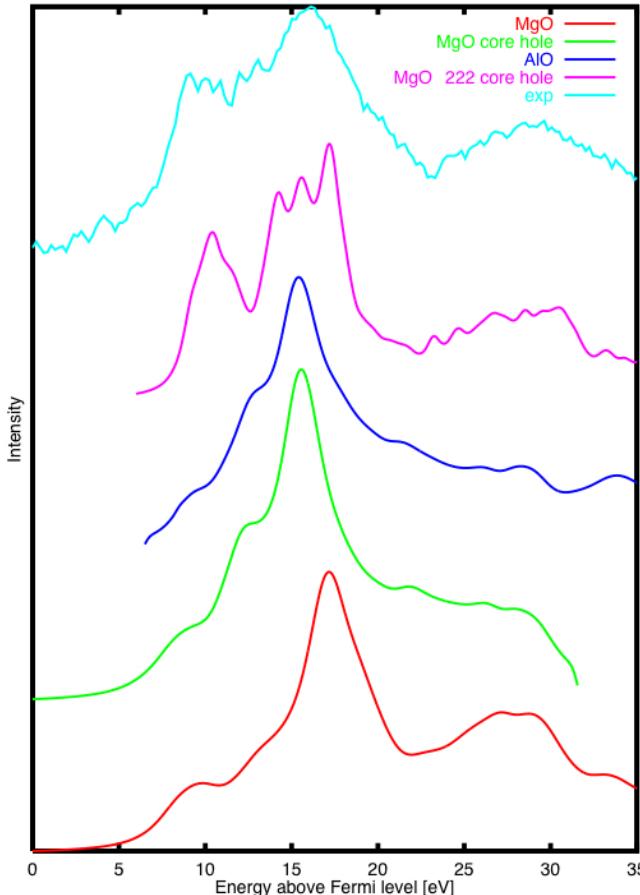


Different ways of treating the core hole within WIEN2k

- No core hole (= ground state, sudden approximation)
- Z+1 approximation (e.g., replace C by N)
- Remove 1 core electron, add 1 electron to conduction band
- Remove 1 core electron, add 1 electron as uniform background charge
- Fractional core hole: remove between 0 and 1 electron charge (e.g. 0.5)
- *You may still get a bad result – correct treatment requires a more advanced theory, e.g. BSE treats electron-hole interaction explicitly (gold standard).*

Core hole calculations usually require a supercell !!!

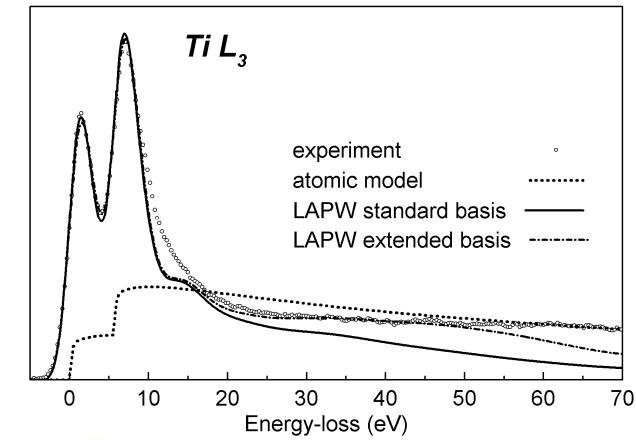
Mg-K in MgO



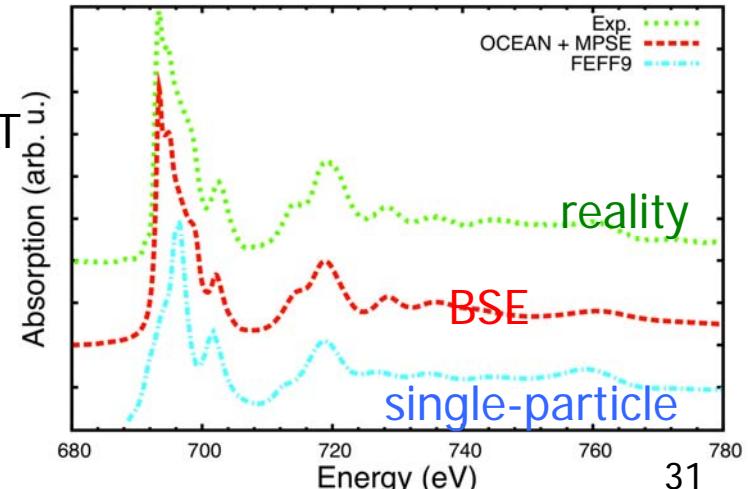
- mismatch between experiment and simulation
- introduction of core hole or $Z+1$ approximation does not help
- interaction between neighbouring core holes
→ core hole in a supercell

Challenges of WIEN2k

1. Basis set only meant for limited energy range :
 - forget about EXAFS/EXELFS
 - sometimes adding a LO (case.in1) with a high linearization energy of 2.0 or 3.0 improves description of high-energy states.

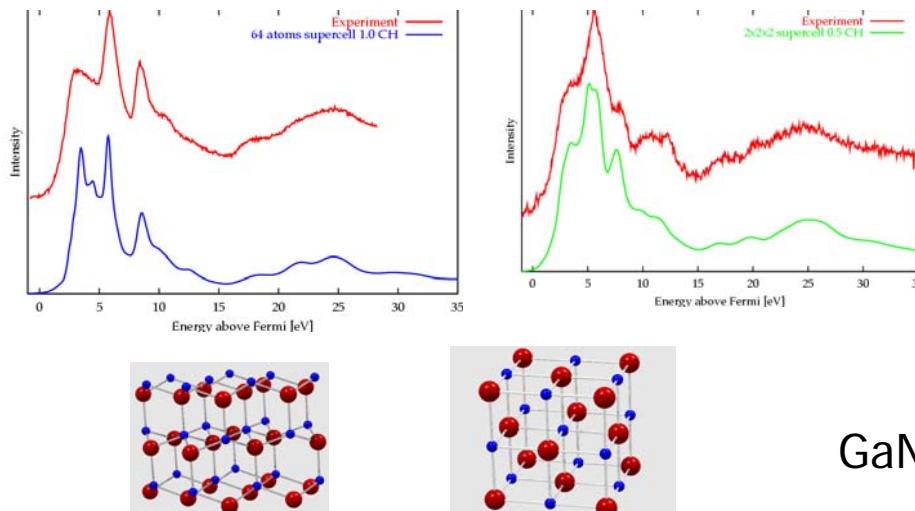


2. Sometimes Final State Rule (core hole) DFT just isn't good enough and you need Bethe-Salpeter (BSE) calculations
 - codes : OCEAN, AI2NBSE, Exc!ting, "BSE"
 - much more expensive
 - not as "polished" as DFT
 - gets L3/L2 ratios right



Challenges of WIEN2k

3. Core hole supercell size can be hard to converge.
- size of the cell
 - how much charge to remove?
 - optimal treatment can differ between similar materials; or even different edges in same material



S. Lazar, C. Hébert, H. W. Zandbergen
Ultramicroscopy 98, 2-4, 249 (2004)

TAILLEFUMIER, CABARET, FLANK, AND MAURI

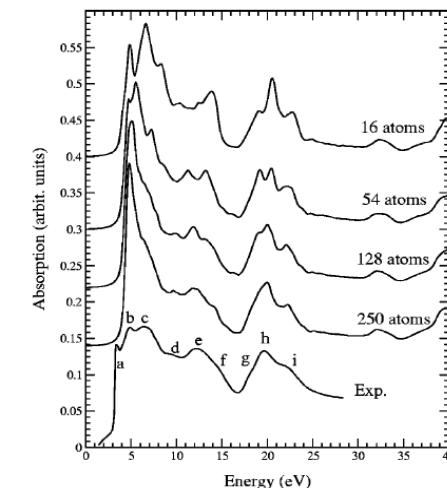
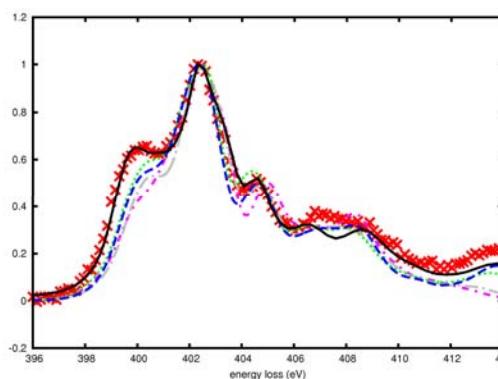
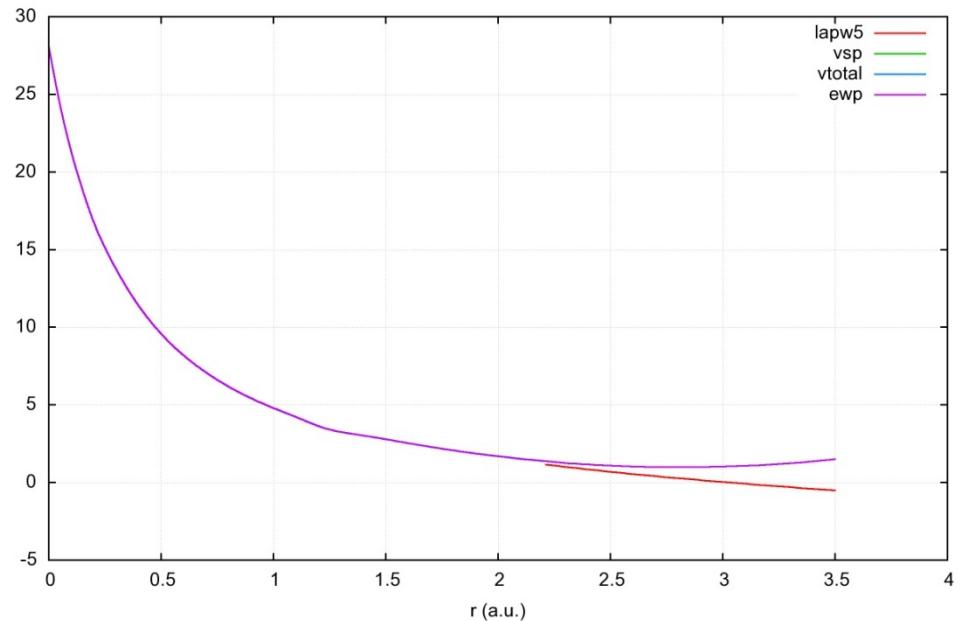
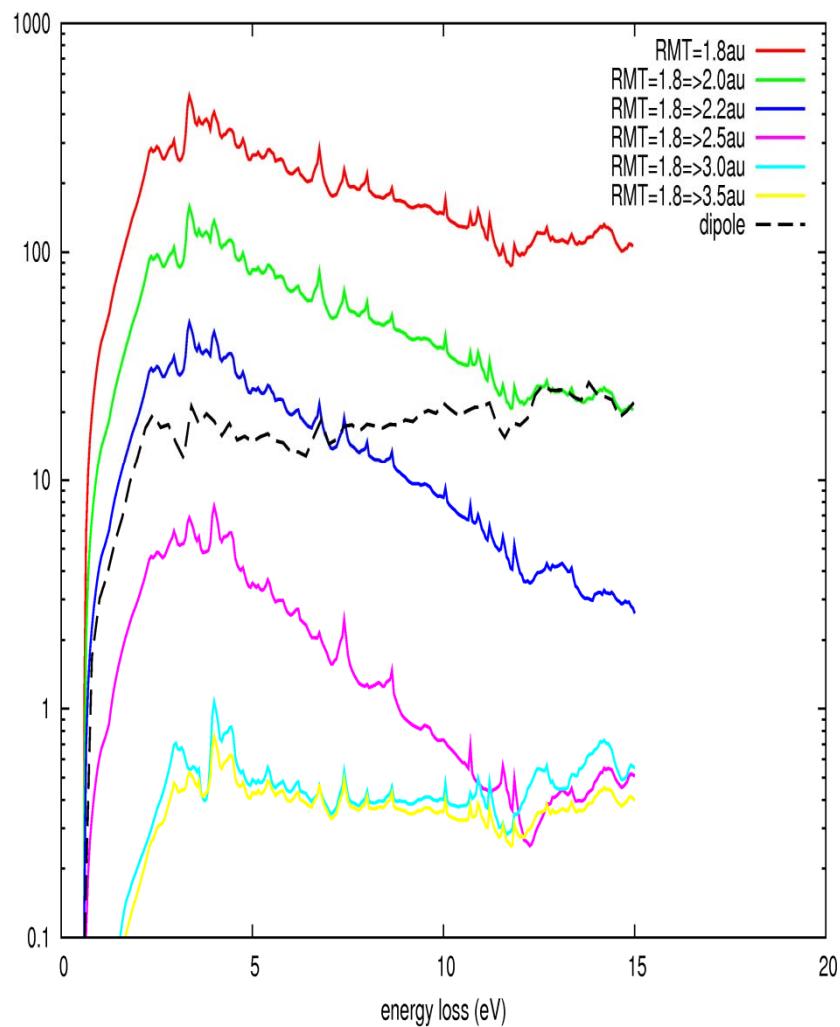


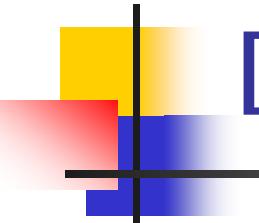
FIG. 1. Calculated C K-edge X-ray absorption spectra in diamond for different supercell sizes, compared with experimental data (from Ref. 42). A 286.1 eV shift was applied to the experimental



Challenges of WIEN2k

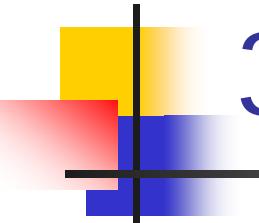


4. Killing artifacts (unphysical monopoles) by “extending the RMT”



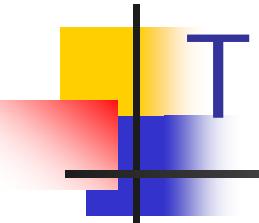
Documentation

- WIEN2k Users Guide!
- C. Hebert, Practical aspects of calculating EELS using the WIEN2k code, Ultramicroscopy, 2007
- Jorissen, Hebert & Luitz, submitting
(http://leonardo.phys.washington.edu/feff/papers/dissertations/thesis_jorissen.pdf - Kevin's Ph.D. thesis)



3. Hands-on exercises

1. XAS of K edge of Cu.
2. averaged EELS of N K edge of GaN.
3. orientation sensitive, in-plane and out-of-plane EELS of N K edge of GaN.
4. core hole calculation for Cu K-edge XAS & compare.
5. initialize a $2 \times 2 \times 2$ supercell for TiC or TiN core hole EELS calculation.
6. Be K edge. Find the error.



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