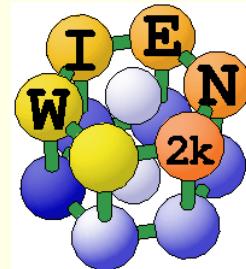


## The FP-LAPW and APW+lo bandstructure methods as implemented in WIEN2k

Peter Blaha

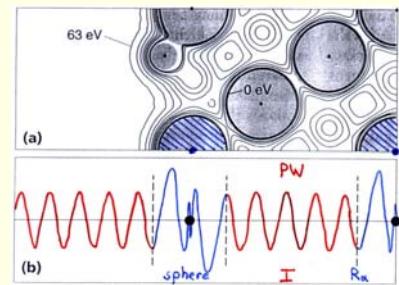
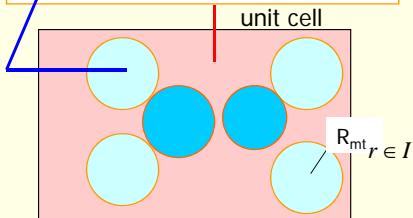
Institute of Materials Chemistry  
TU Wien

[http://wien2k.at/events/vicom\\_wien2k.pdf](http://wien2k.at/events/vicom_wien2k.pdf)



## APW Augmented Plane Wave method

The unit cell is partitioned into:  
atomic spheres  
Interstitial region



Basisset:

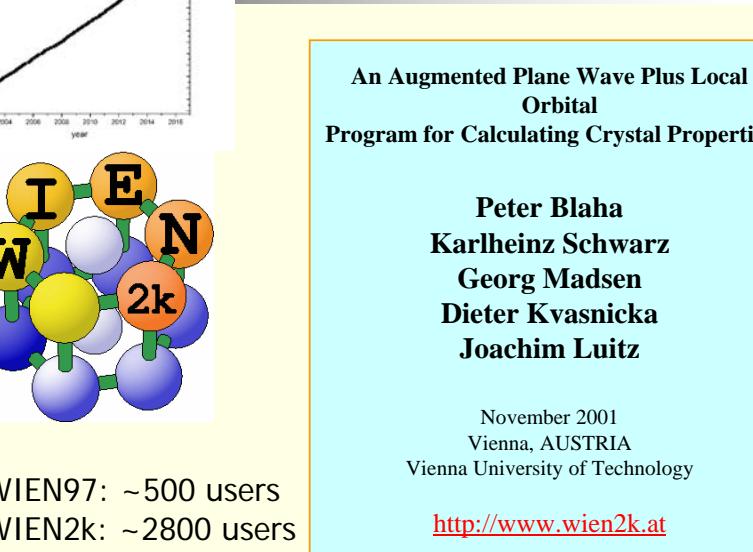
$$\text{PW: } e^{i(\vec{k} + \vec{K}) \cdot \vec{r}}$$

Atomic partial waves

$$\sum_{\ell m} A_{\ell m}^K u_{\ell}(r', \varepsilon) Y_{\ell m}(\hat{r}')$$

join

$u_{\ell}(r, \varepsilon)$  are the numerical solutions of the radial Schrödinger equation in a given spherical potential for a particular energy  $\varepsilon$   
 $A_{\ell m}^K$  coefficients for matching the PW



The slide features a blue header bar with the text "WIEN2k software package" and the TU Wien logo. Below the header is a graph showing the exponential growth of WIEN2k licenses from 2002 to 2016. The graph has "Year" on the x-axis and "Licenses" on the y-axis, ranging from 0 to 2800. A legend indicates "WIEN2k licenses". To the left of the graph is a 3D molecular model composed of spheres labeled "W", "E", "N", and "2k". Below the graph, text compares WIEN97 (~500 users) and WIEN2k (~2800 users). The main title "An Augmented Plane Wave Plus Local Orbital Program for Calculating Crystal Properties" is centered above the author list. The authors listed are Peter Blaha, Karlheinz Schwarz, Georg Madsen, Dieter Kvasnicka, and Joachim Luitz. The date "November 2001" and location "Vienna, AUSTRIA" are at the bottom, along with the URL "http://www.wien2k.at".

WIEN2k software package

TU  
WIEN

WIEN2k licenses

Year

Licenses

An Augmented Plane Wave Plus Local Orbital Program for Calculating Crystal Properties

Peter Blaha  
Karlheinz Schwarz  
Georg Madsen  
Dieter Kvasnicka  
Joachim Luitz

November 2001  
Vienna, AUSTRIA  
Vienna University of Technology

<http://www.wien2k.at>

WIEN97: ~500 users  
WIEN2k: ~2800 users

The diagram illustrates the WIEN2k software architecture, showing the flow of data and processes from input generation to the final output.

**Input Generation:**

- init\_lapw** (blue box):
  - step-by-step or batch initialization
  - symmetry detection ( $F$ ,  $I$ ,  $C$ -centering, inversion)
  - input generation with recommended defaults
  - quality (and computing time) depends on  $k$ -mesh and  $R.Kmax$  (determines #PW)
- run\_lapw** (blue box):
  - scf-cycle (optional position optim.)
  - optional with SO and/or LDA+U
  - different convergence criteria (energy, charge, forces)
- save\_lapw tic\_gga\_100k\_rk7\_volo** (red box):
  - cp case.struct and clmsum files,
  - mv case.scf file
  - rm case.broyd\* files

**Process Flow:**

- init\_lapw** leads to **SGROUP** and **SYMMETRY**.
- SGROUP** leads to **STRUCTURE** (input files).
- SYMMETRY** leads to **LSTART** (atomic calculation,  $R.PW = 1$ ), which then leads to **atomic densities** (input files) and **KGEN** (k-mesh generation).
- LSTART** leads to **DSTART** (superposition of atomic densities) and **P**.
- DSTART** leads to **LAPW0** (Poisson, LDA,GGA,  $V_{xc} + V_{xc}$ ).
- LAPW0** leads to **ORB** (LDA+U, DP potentials).
- ORB** leads to **LAPW1** ( $[-\nabla^2 + v] \Psi_1 = E_1 \Psi_1$ ).
- LAPW1** leads to **LAPW2** ( $\Psi_{tot} = \sum_i \Psi_i$ ,  $E_{tot}$ ).
- LAPW2** leads to **LAPWSO** (add spin-orbit interaction).
- LAPWSO** leads to **LAPWDM** (Calc. density matrix).
- LAPWDM** leads to **LCORE** (atomic calculation,  $H_{tot} = E_{tot} \Psi_{tot}$ ,  $R_{tot}$ ,  $E_{tot}$ ).
- LCORE** leads to **MIXER** ( $\Psi_{tot} = \Psi_{tot} \otimes \Psi_{tot} + \Psi_{tot}^*$ ).
- MIXER** leads to **STOP** (yes/no).
- STOP** leads to **converged?** (yes/no).
- converged?** leads back to **LAPW0** if no, or to **STOP** if yes.

**WIEN2k and the w2web GUI**

Session: TiC  
/area51/pbla/lapw/2005-june/TiC

**StructGen™**

You have to click "Save Structure" for changes to take effect!

Title: TiC  
Lattice:  
Type: F

Spacegroups from Bilbao Cryst Server

Lattice parameters in A  
a=4.328000038 b=4.328000038 c=4.328000038  
g=90.000000 β=90.000000 γ=90.000000

Inequivalent Atoms: 2

Atom 1: Ti      Z=22.0      RMT=2.0000      remove atom  
Pos 1: x=0.0000000 y=0.0000000 z=0.0000000 remove add position

Atom 2: C      Z=6.0      RMT=1.9000      remove atom  
Pos 1: x=0.5000000 y=0.5000000 z=0.5000000 remove add position

**exercises: NiO an AFM insulator**

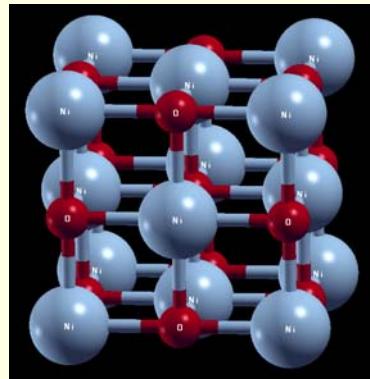
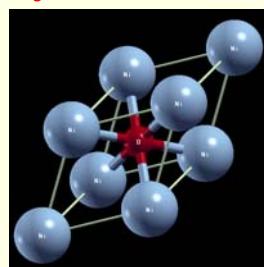
- open the userguide in a browser: [http://wien2k.at/reg\\_user/textbooks](http://wien2k.at/reg_user/textbooks)
- open 2 windows and connect to the vsc3 frontend
  - `ssh -X training@vsc3.vsc.ac.at` (Vicom\_2017)
- `salloc -J pc##` (## is your PC number: 01-29)
- `srun hostname` (find the allocated compute node)
  - `ssh -X nAA-BBB` (see previous line on your screen, on all windows)
- on ALL windows: `cd ##/wien2k` (see label on your screen)
  - since we all use the same account , it is ESSENTIAL that you create data only in YOUR „home-directory” !!
- the „text-version“ of the instructions (for „cut and paste“) can be opened using
  - `$EDITOR ~/bla/bla/wien2k_exercises.txt &` (or use less/ vi / emacs)



## NiO: nonmagnetic calculation



- mkdir NiO-nm; cd NiO-nm
- makestruct\_lapw # define the NaCl structure of NiO
  - lattice:  $F$  (one does not need to know the spacegroup)
  - lattice parameter: 4.186 Ang
  - Ni (0,0,0)
  - O (0.5,0,0)
- cp init.struct NiO-nm.struct
- xcrysden --wien\_struct .



primitive and conventional cell



## NiO-nm: initialization and scf



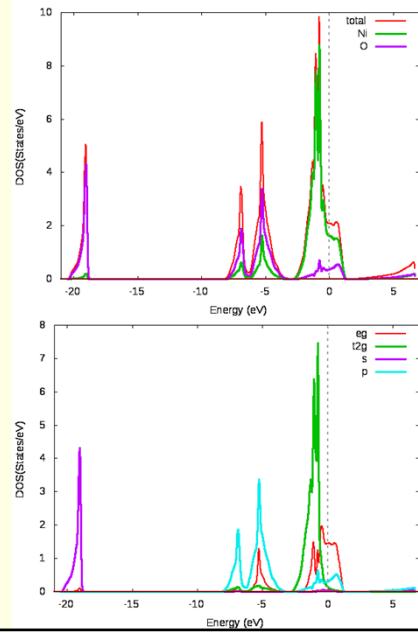
- init\_lapw -h # help switch to see all options
- init\_lapw -b -numk 300 # batch initialization with 6x6x6 kmesh  
# what are the valence and core states ?
- run\_lapw # scf cycle with default convergence
- save\_lapw pbe # save a calculation
- # check scf convergence:
  - grep :ENE pbe.scf # observe convergence and warnings
  - grep :WAR pbe.scf # linearization warning for state above EF
  - grep :FER pbe.scf
  - grep :GAP pbe.scf # metal !!
  - grep :DIS pbe.scf # charge convergence
  - grep :CTO001 pbe.scf # charge in Ni sphere (~constant, Ni<sup>2+</sup>, 3d<sup>8</sup>)
  - grep :CTO002 pbe.scf # O charge (increases, O<sup>2-</sup>, Ni-4s<sup>2</sup> → O-2p<sup>6</sup>)



## DOS (total, Ni-d<sub>eg/t2g</sub>, O-s,p)



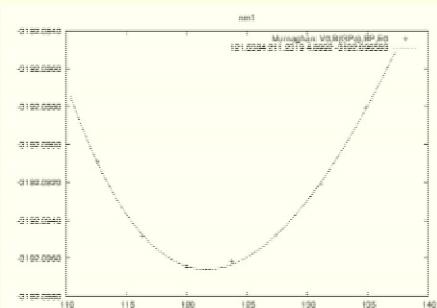
- **x lapw2 -qtl**
- **configure\_int\_lapw**
  - *select: total*
  - *1 (tot,d-eg,d-t2g)*
  - *2 (tot,s,p)*
  - *E-min/Emax: -1.0 to 1.0 Ry*
- **x tetra**
- **dosplot2**
  - *produce 2 plots:*
  - *total, Ni-tot, O-tot*
  - *Ni-eg,t2g,O-s,p)*



## Volume optimization



- **x optimize** # create a series of struct files with different volume
  - *select option 1 (volume) and 7 cases: -9 -6 -3 0 3 6 9 % change of vol*
- **./optimize.job** # run scf for all volumes.
  - *this takes 2-3 min and while this is running, you may continue in the other window with the ferromagnetic calculation and come back later*
- **grepine :ene '\*scf' 1** # get E-tot for all volumes
- **eplot -a default** # fit EOS and plot E vs. vol
  - *what is  $a_0$  (compare to experiment) and  $B_0$ ?*





## NiO: ferromagnetic calculation



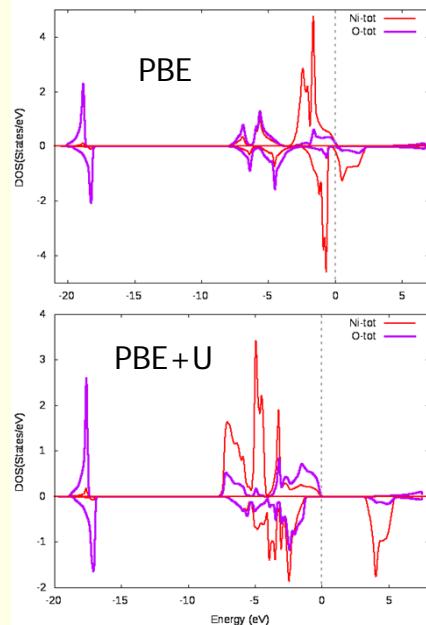
```
■ cd ..; mkdir NiO-fm; cd NiO-fm  
■ cp ../NiO-nm/NiO-nm.struct NiO-fm.struct      #copy nm struct file  
■  
■ init_lapw -b -sp -numk 300      # spin-polarized initialization  
■ runsp_lapw                      # spin-polarized scf cycle  
■ save_lapw pbe  
■ # check convergence and compare with NiO-nm  
■ grep :ENE *.scf      # fm or nm lower in energy ?  
■ grep :GAP *.scf      # still metallic, (exp. gap ~4eV)  
■ grep :MMT *.scf      # total spin magnetic moment / cell  
■ grep :MMI001 *.scf    # exp. moment: ~2uB
```



## DOS (total, Ni-d<sub>eg/t2g</sub>, O-s,p)



```
■ x lapw2 -qtl -up  
■ x lapw2 -qtl -dn  
■ configure_int_lapw  
  ■ select: total  
  ■ 1 (tot,d-eg,d-t2g)  
  ■ 2 (tot,s,p)  
  ■ E-range = -1.0 / 1.0  
■ x tetra -up / -dn  
■ dosplot2 -up  
  ■ produce a spin-pol. plot:  
  ■ Ni-tot, O-tot up+dn
```





## NiO: fm PBE+U calculation



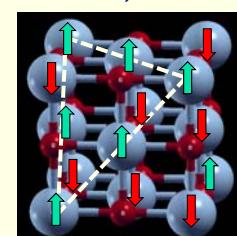
- `x orb -up`
- `x lapwdm -up` # create input templates
- `$EDITOR NiO-fm.indm` # density matrix for only 1atom (Ni, l=2; delete 2<sup>nd</sup> atom)
- `$EDITOR NiO-fm.inorb` # only 1 atom (Ni), SIC-method, U= 7eV
  
- `runsp_lapw -orb` # spin-polarized scf cycle with GGA+U
- `save_lapw pbe+u`
- # compare with fm calculation
- `grepline :GAP '* .scf' 2` # shows last 2 :GAP lines of all scf files
- `grepline :MMI001 '*scf' 1` # observe creation of gap and larger moment
  
- plot DOS in the same way as for PBE calculation

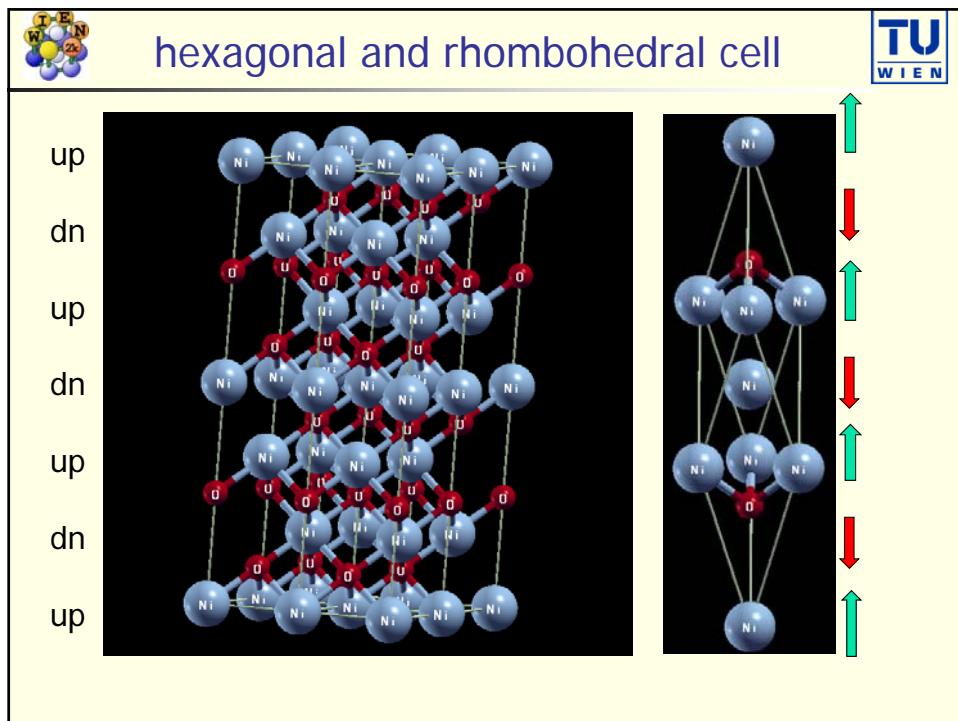


## NiO: AFM-II (fm Ni(111) layers)



- `cd ..; mkdir NiO-afm; cd NiO-afm`
- `cp .. /NiO-nm/NiO-nm.struct fcc.struct` #copy nm struct file as start
- create a supercell with 2 Ni and O atoms along (111)-direction
- `octave` # a „free“ matlab
- `helpstruct` # list all structeditor commands
- `help loadstruct` # help for specific command
- `s=loadstruct("fcc.struct");`
- `sp=makeprimitive(s);` # create primitive (rhombohedral) cell
- `su=makesupercell(sp,[1 1 0; 1 0 1; 0 1 1]);`
- `showstruct(su)`
- `savestruct(su,"NiO-afm.struct")`
- `quit`






 AFM-II NiO
 

```

■ less NiO-afm.struct      # observe the 4 atoms in R cell
■ x sgroup                 # run spacegroup program
■ less NiO-afm.struct_sgroup # back to original NaCl structure
■ $EDITOR NiO-afm.struct   # „label Ni1 and Ni2“ (overwrite mode).
  This directs sgroup to treat Ni1 and Ni2 as different atoms
■ x sgroup
■ less NiO-afm.struct_sgroup # 3 non-equivalent atoms !
■ cp NiO-afm.struct_sgroup NiO-afm.struct # take this struct file
■ instgen_lapw –ask          # starting spin-structure: u, d, n
■ init_lapw –b –sp –numk 100 # (4x4x4 k-mesh)
■ runsp_lapw                  # spin-polarized scf cycle
■ save_lapw pbe
    
```

## AFM NiO: convergence and DOS



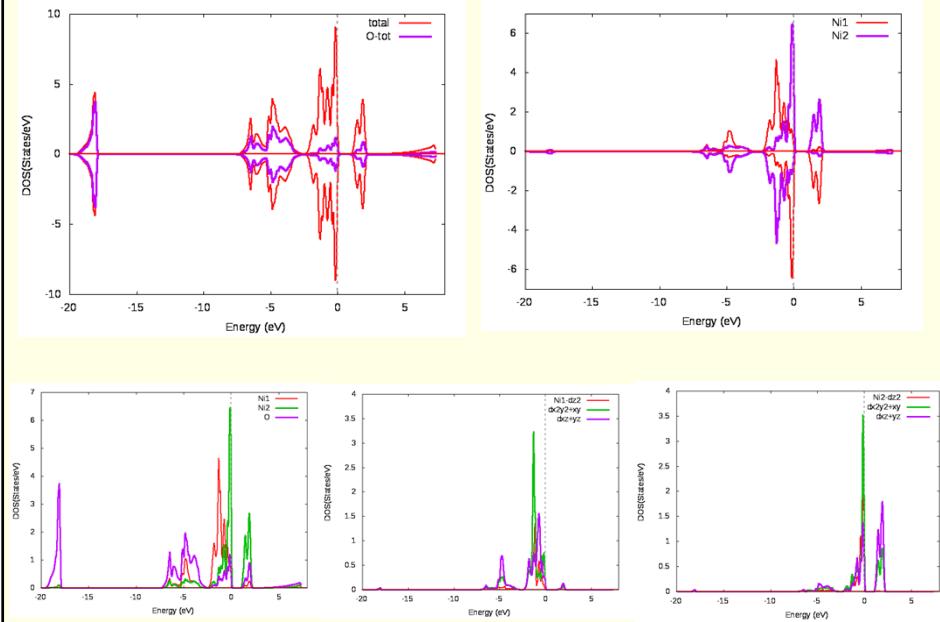
```

■ # check convergence and compare with NiO-nm, NiO-fm
■ grep :ENE *.scf      # afm, fm or nm lower in energy ?
■ grep :GAP *.scf      # small gap opened, (exp. gap ~4eV)
■ grep :MMT *.scf      # total spin magnetic moment / cell
■ grep :MMI001 *.scf   # small moment (exp. ~2uB)

■ # plot DOS (observe „symmetry“ between up and dn-DOS)
■ x lapw2 -qtl -up /-dn
■ configure_int_lapw      # select total and all „meaningful“ splittings
                            (they are automatically symmetry adapted), E-range
■ x tetra -up / -dn
■ dosplot2 -up           # Ni1/2-up/dn; d-split (eg/t2g does not exist in this
                            coordinate system); Ni1,Ni2,O-tot up

```

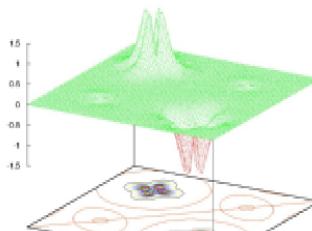
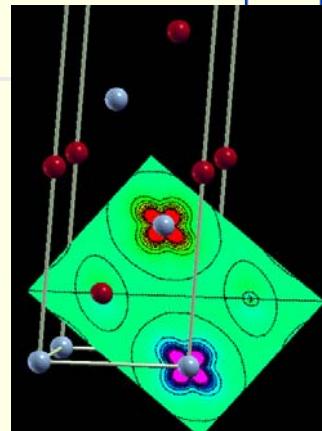
## AFM NiO: PBE DOS





## spin-density

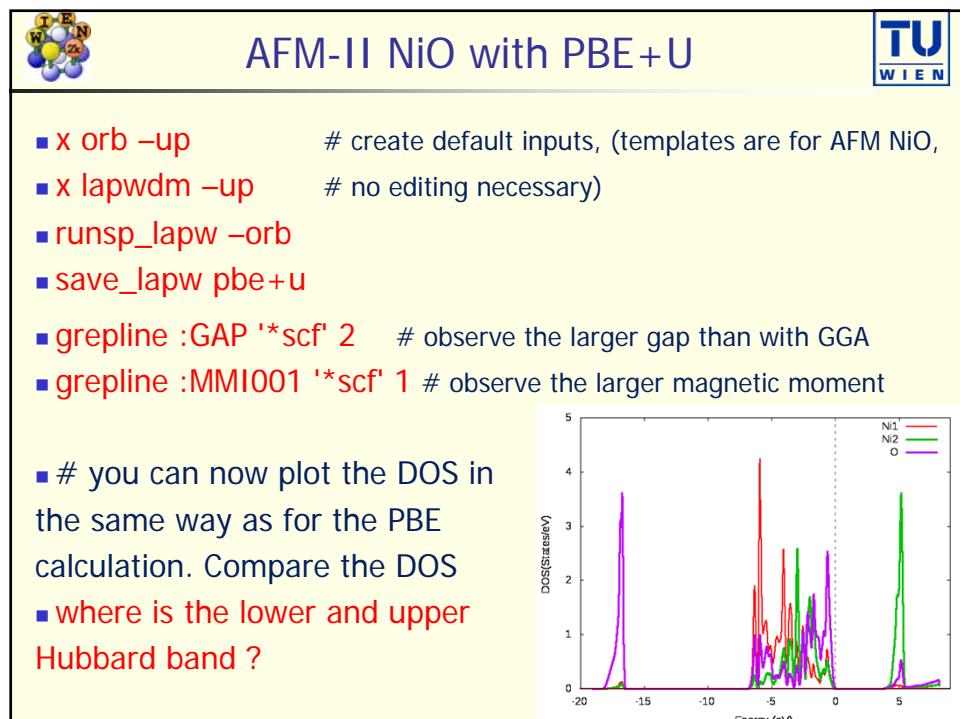
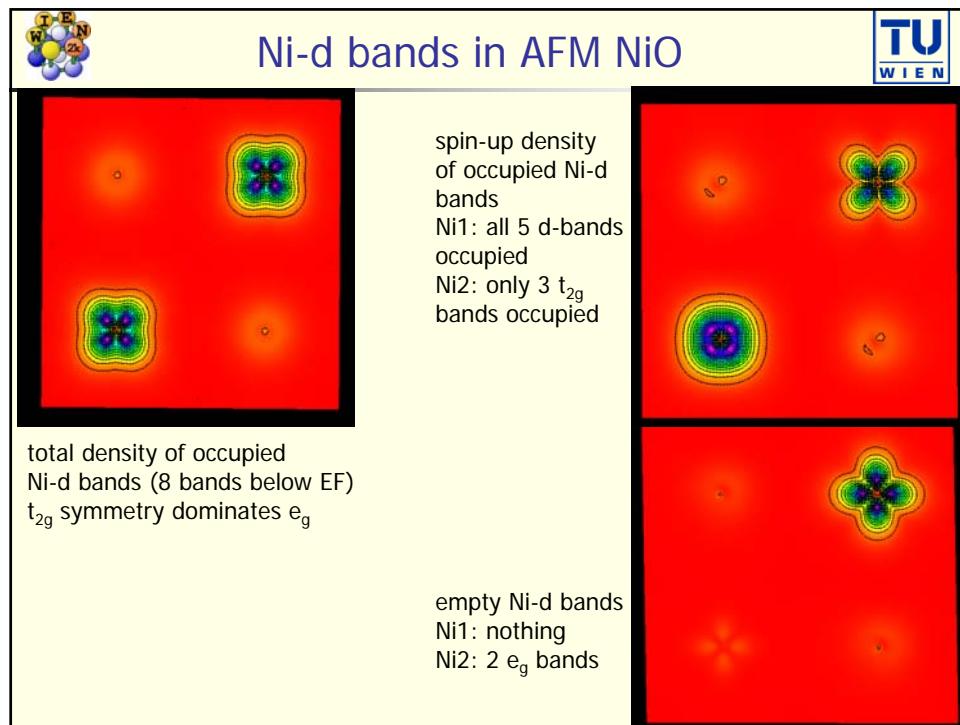
- `xcrysden --wien_density .`
  - 2D plot
  - 80 points (in first line)
  - 3 atoms spanning rectangular „fcc“-100-plane (2Ni at different z-values)
  - enlarge plane by 0.5
  - submit and change „ADD“ to „SUB“
  - plot with „rainbow“, thermometer, ranges: -0.4/0.40, small atoms, ball/stick ratio=0
  
- `rhoplot` # 3D-plot



## density of occupied and empty Ni-d states



- `grep band NiO-afm.output2up` # find the energy-ranges (xxx and yyy) around EF to plot occupied (8 bands) and empty (2 bands) Ni-d states
- `x lapw2 -up -emin xxx` # calculate  $\rho_{up}$  for states from xxx to EF
- `x lapw2 -dn -emin xxx` # calculate  $\rho_{dn}$  for states from xxx to EF
- `$EDITOR NiO-afm.in5` # change "SUB" to "ADD"
- `x lapw5 -up` # generate density in plane
- `rhoplot / xcrysden --wien_renderdensity .` # plot total density of occupied Ni-d band - which symmetry is present around Ni
- `rm NiO-afm.clmvaldn` # remove spin-dn density
- `x lapw5 -up`
- `rhoplot / xcrysden --wien_renderdensity .` # plot spin-up density of occupied Ni-d band - which symmetry is present around Ni
  
- `x lapw2 -all xxx yyy -up` # calculate spin-up density for unoccupied Ni-d band
- `x lapw5 -up`
- `xcrysden --wien_renderdensity .` # plot spin-up density of unoccupied Ni-d band - symmetry around Ni ?





## Compare the different calculations



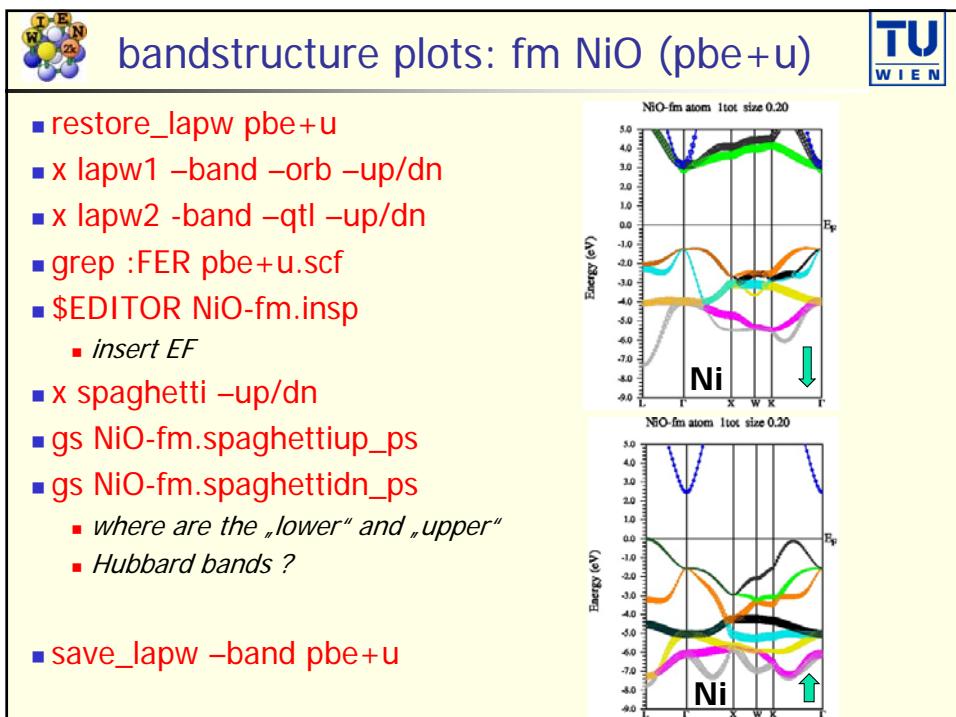
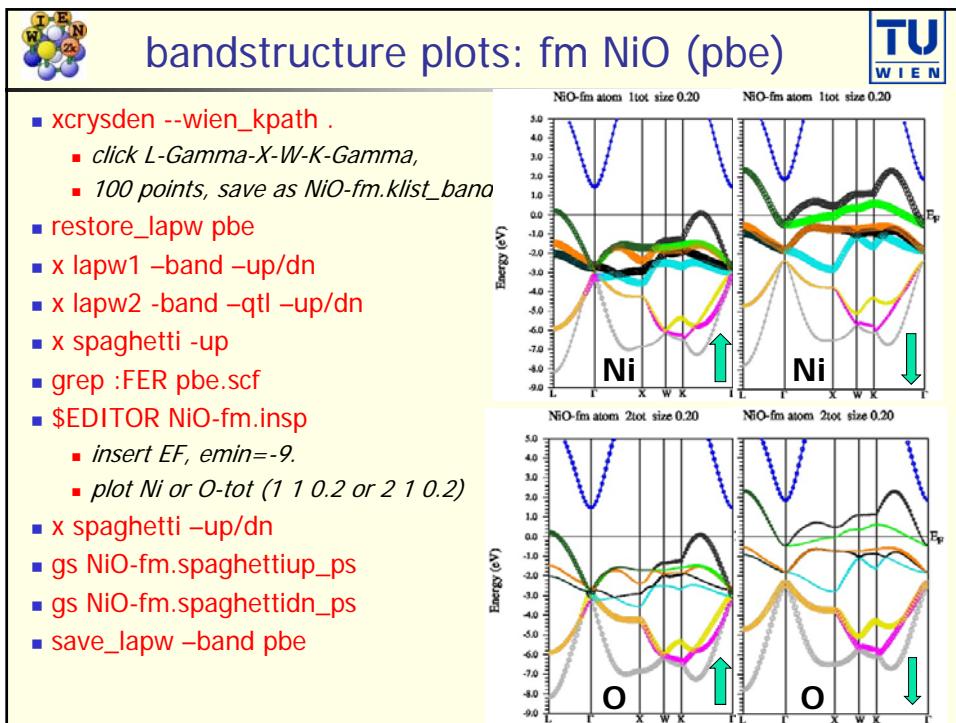
- `cd ..`
- `grepline :ENE '*' /pbe.scf' 1` # which case has the lowest energy, what are the differences ?
- `grepline :ENE '*' /pbe+u.scf' 1` # which case has the lowest energy, what are the differences ? What are the differences between PBE and PBE+U ?
- # from an analysis of total energies for different spin structures (including other AFM settings) a Heisenberg model can be set up and the exchange parameters  $J_i$  could be determined.
- `grepline :GAP '*' /pbe.scf' 2` # which case has a gap in PBE ?
- `grepline :GAP '*' /pbe+u.scf' 2` # which case has a gap in PBE+U ?



## FM NiO with GGA+U and spin-orbit coupling



- `cd NiO-fm`
- `initso_lapw` # define magnetization direction along (001), use defaults; continue setup with "spin-polarization": symmetso will detect the symmetry break and create a new structure, commit all changes
- `runsp_lapw -orb -so` # scf cycle with PBE+U and spin-orbit
- `save_lapw pbe+u_so`
- `grepline :GAP '*' /scf' 2` # compare the gaps
- `grepline :MMI001 '*' /scf' 1` # compare the spin magnetic moments
- `grepline :ORB001 '*' /scf' 1` # get the orbital magnetic moment (only in the so-calc.)
- get the DOS as above (use `x_lapw2 -up -so -qtl`)

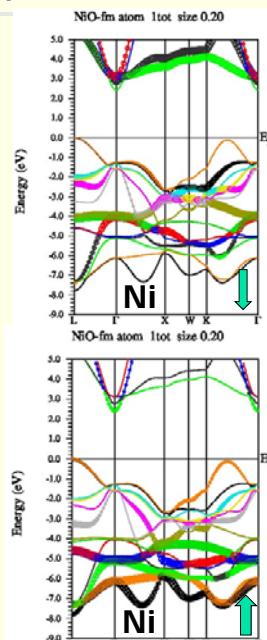




## bandstructure: fm NiO (pbe+u+so)



- restore\_lapw pbe+u\_so
- x lapw1 -band -up/dn
- x lapwso -up -orb
- x lapw2 -band -qt1 -so -up/dn
- grep :FER pbe+u\_so.scf
- \$EDITOR NiO-fm.insp
  - Insert EF
- x spaghetti -so -up/dn
- gs NiO-fm.spaghettiup\_ps
- gs NiO-fm.spaghettidn\_ps
  - what is the difference to calc. without spin-orbit ?
- save\_lapw -band pbe+u\_so



## other DFT options



- Tran-Blaha modified Becke-Johnson potential
  - a meta-GGA potential giving gaps of GW quality. XES spectra ok.
  - fast, but no total energy ! init\_mbj\_lapw (see UG)
- EECE (onsite hybrid-DFT for correlated electrons only).
  - fast, similar to GGA+U, good gaps with proper HF fraction (see UG)
- Hybrid-DFT (PBE0, HSE, ...)
  - rather time consuming, run in k-parallel (or mpi-) mode, good gaps with proper HF fraction. init\_hf\_lapw (see UG)



## DOS of NiO

- mBJ DOS agrees with
  - XPS/BIS
  - Ni-XES, O-XES
- LDA+U gives similar gap, but cannot explain XES
- PBE0: gap too large

